

Thermodynamique des Processus Irréversibles
Quiberon, 16-22 sept. 2018

7ème école d'été de mécanique théorique

Effets de couplage et effets dissipatifs accompagnant la déformation des matériaux solides

(3^{ème} partie)

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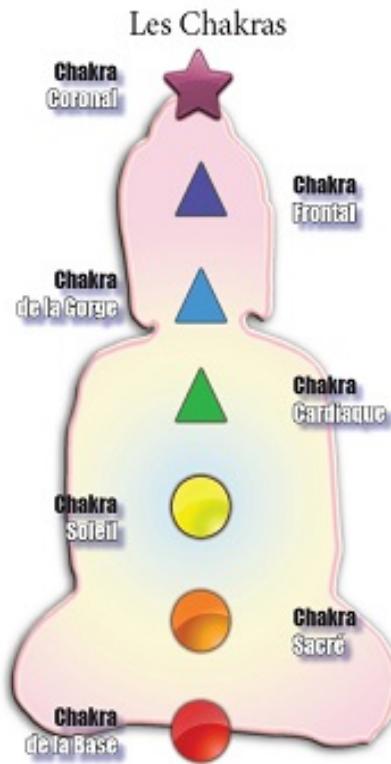


LABORATOIRE DE MÉCANIQUE ET GÉNIE CIVIL - UM/CNRS

$$I_\varepsilon(\varphi) = \int_{\Omega} V\varphi(\boldsymbol{\eta}) \, d\boldsymbol{\eta} \quad \lim_{\varepsilon \rightarrow 0^+} I_\varepsilon(\varphi) = W(\cdot, F) = \infty$$

Programme

- 1 - Cadre thermomécanique et bilan d'énergie
(TPI-MSG)
- 2 - Quelques éléments rhéologiques à la sauce MSG
(d'une vision mécanique à vision thermomécanique)
- 3 - Analyse expérimentale des bilans d'énergie
(imagerie quantitative)
- 4 - Effet du temps : couplage thm et/ou viscosité ?
(interaction forte et/ou irréversibilité)
- 5 - Effet dissipatif dans les métaux
(fatigue : HCF & VHCF)



BILAN ENERGETIQUE - GRATUIT

*Laissez-vous surprendre par la pertinence
de ce bilan offert et découvrez comment :*

- ★ *Avoir plus d'énergie !*
- ★ *Augmenter votre taux vibratoire !*
- ★ *Vivre en pleine forme !*

Bilan d'énergie :
le retour !



$$I_\varepsilon(\varphi) = \int_{\Omega} V(\varphi(\boldsymbol{x})) \, d\boldsymbol{x} \quad \lim_{\varepsilon \rightarrow 0^+} I_\varepsilon(\varphi) = W(\cdot, F) = \infty$$

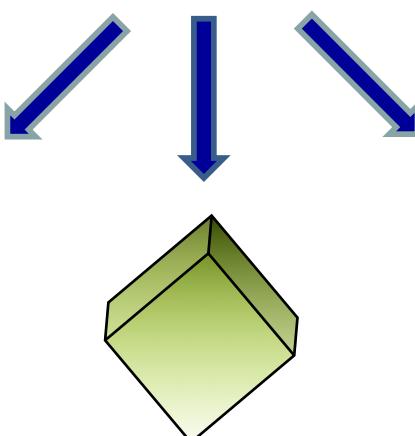
Energy balance : reminder (I)

$$\begin{aligned} W_{\text{def}}^\bullet &= \sigma : \dot{\varepsilon} = \sigma^r : \dot{\varepsilon} + \sigma^{ir} : \dot{\varepsilon} \\ &= \underbrace{\sigma^r : \dot{\varepsilon} + A_\alpha \cdot \dot{\alpha}}_{W_e^\bullet + W_s^\bullet} + d_1 \end{aligned}$$



$(\dots)^\bullet$ path-dependence

W_e^\bullet : rate of elastic energy



d_1 : intrinsic dissipation

W_s^\bullet : rate of stored energy



... incomplete balance !!

$$I_\varepsilon(\varphi) = \int_{\Omega} V(\varphi(\boldsymbol{x})) \, d\boldsymbol{x} \quad \lim_{\varepsilon \rightarrow 0^+} I_\varepsilon(\varphi) = W(\cdot, F) = \infty$$

Energy balance : reminder (II)

- rate of internal energy

$$\rho \dot{e} = \rho C \dot{T} + (\sigma^r : \dot{\epsilon} + A \cdot \dot{\alpha}) - (T \sigma_{,T}^r : \dot{\epsilon} + T A_{,T} \cdot \dot{\alpha})$$

$$= \rho C \dot{T} + w_e^\bullet + w_s^\bullet - w_{thc}^\bullet$$

« thc » = thermomechanical couplings

- heat equation

$$\rho C \dot{T} + \text{div} \mathbf{q} = \sigma^r : \dot{\epsilon} - \mathbf{A} \cdot \dot{\alpha} + T \sigma_{,T}^r : \dot{\epsilon} + T \mathbf{A}_{,T} \cdot \dot{\alpha} + r_e$$

d_1



kinematics
required

- comments

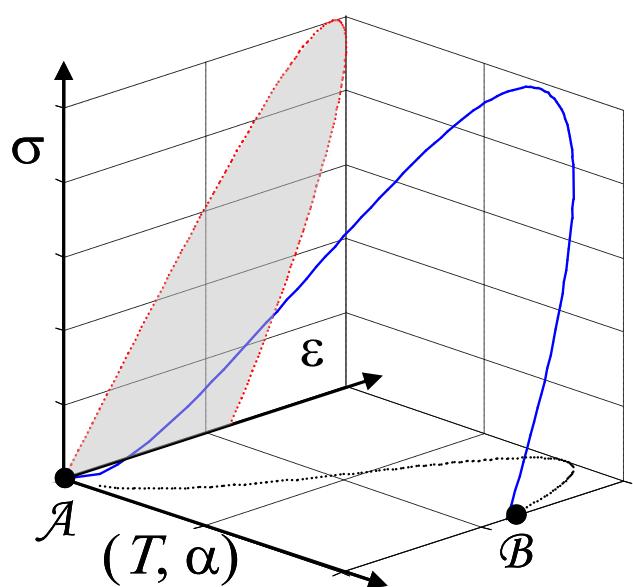
C.1: C specific heat

C.2: $q = -k \cdot \text{grad} T$

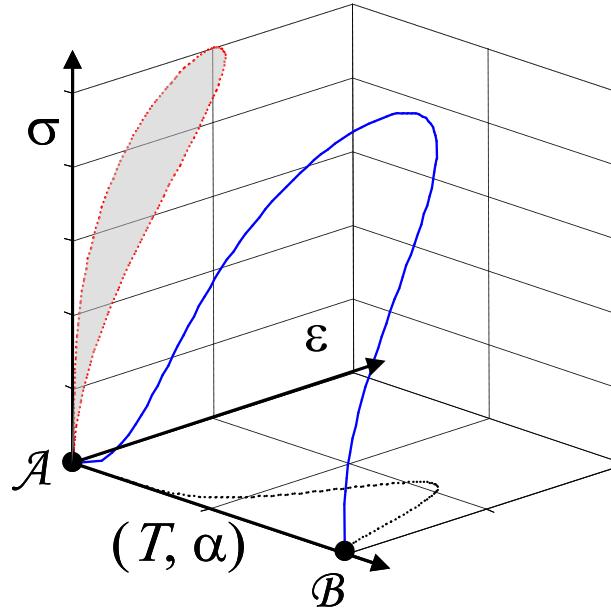
C.3: $\dot{T} = \frac{\partial T}{\partial t} + \mathbf{v} \cdot \nabla T$

$$I_\varepsilon(\varphi) = \int_{\Omega} V(\varphi) \, dx + \int_{\Omega} \nabla \varphi \cdot \nabla W(\varphi) \, dx$$

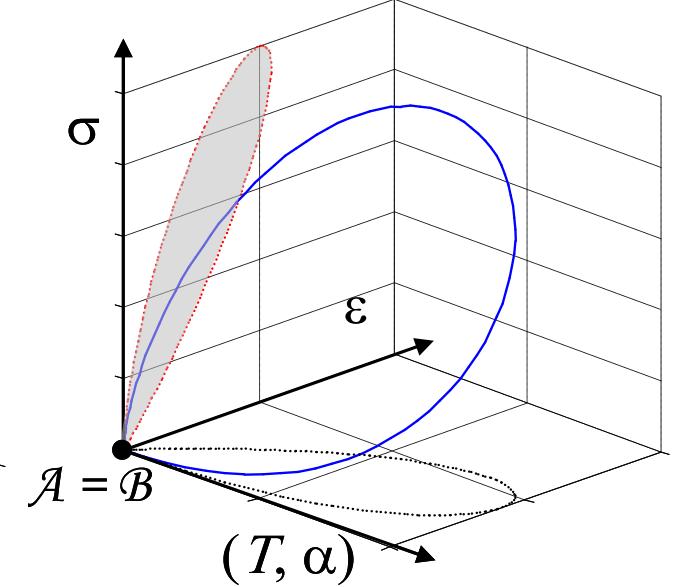
Focus on a load-unload cycle



(i) $\mathcal{A} \neq \mathcal{B}$



(ii) $\varepsilon_{\mathcal{A}} = \varepsilon_{\mathcal{B}}$



(iii) $\mathcal{A} = \mathcal{B}$

$$(i) \quad w_{\text{def}} = \int_{t_{\mathcal{A}}}^{t_{\mathcal{B}}} \sigma : \dot{\varepsilon} \, dt = \int_{t_{\mathcal{A}}}^{t_{\mathcal{B}}} d_1 \, dt + \int_{t_{\mathcal{A}}}^{t_{\mathcal{B}}} (\rho \dot{e} - \rho C \dot{T} + w_{\text{thc}}^{\bullet}) \, dt$$

(ii) Hysteresis loop : $w_{\text{def}} = A_h$ (for uniaxial loading)

(iii) Load-unload cycle = thermodynamic cycle

$$w_{\text{def}} = \int_{t_{\mathcal{A}}}^{t_{\mathcal{B}}} d_1 \, dt + \int_{t_{\mathcal{A}}}^{t_{\mathcal{B}}} w_{\text{thc}}^{\bullet} \, dt$$



$$I_\varepsilon(\varphi) = \int_{\mathbb{R}^d} V(\varphi(x)) dx + \lim_{n \rightarrow \infty} W(\cdot, F) = \infty$$

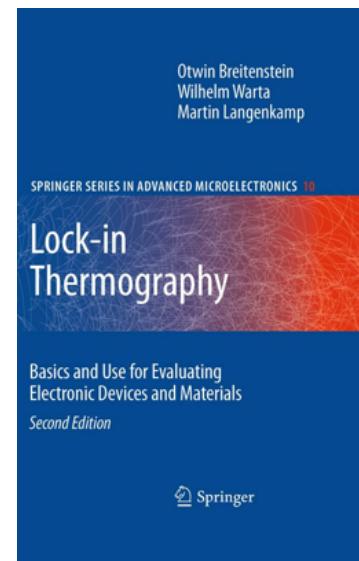
- 4 -

Effet du temps : couplage thm et/ou viscosité ?
(interaction forte et/ou irréversibilité)

$$I_\varepsilon(\varphi) = \int_{\Omega} V_\varepsilon(\varphi(x)) \, dx \quad \text{and} \quad \lim_{\varepsilon \rightarrow 0^+} I_\varepsilon(\varphi) = W(\cdot, F) = \infty$$



TSA : thermoélasticité standard
TSA : analyse thermique des contraintes



Modèle thermoélastique linéaire isotrope homogène

- mécanique : mémoire sélective
- thermodynamique : pas de dissipation intrinsèque
- variables d' état : (θ, ε) , $\theta = T - T_0$ (HPPT $\theta \ll T_0$)



Physique 1 : Superposition des déformations élastiques et thermiques

Version 1 D

$$\varepsilon = \frac{\sigma^r}{E} + \alpha_{th}\theta$$

$$\sigma^r = E(\varepsilon - \alpha_{th}\theta)$$

Version 3 D

$$\varepsilon_{ij} = \frac{1+\nu}{E} \sigma_{ij}^r - \frac{\nu}{E} \sigma_l^r \delta_{ij} + \alpha_{th} \theta \delta_{ij}$$

$$\sigma_{ij}^r = \lambda \varepsilon_l \delta_{ij} + 2\mu \varepsilon_{ij} - (3\lambda + 2\mu) \theta \delta_{ij}$$

Physique 2 : chaleur spécifique constante

$$C_\varepsilon = -T \frac{\partial^2 \Psi}{\partial T^2} \approx -T_0 \frac{\partial^2 \Psi}{\partial T^2} = C_0 \quad \Rightarrow$$

$$\frac{\partial^2 \Psi}{\partial T^2} = -\frac{C_0}{T_0}$$

α_{th} dilatation
 C_ε chal. spé.
 $\varepsilon_l = \text{tr}(\varepsilon)$
 $\varepsilon_{||} = \frac{1}{2} \varepsilon_{ij} \varepsilon_{ij}$

$$I_\varepsilon(\varphi) = \int_{\mathbb{R}^n} V(\varphi(x)) dx \quad \lim_{\varepsilon \rightarrow 0^+} I_\varepsilon(\varphi) = W(\cdot, F) = \infty$$

□ énergie libre

1D $\rho_0 \psi(\theta, \varepsilon) = \frac{1}{2} E(\varepsilon - \alpha_{th} \theta)^2 - \frac{\rho_0 C_0}{2T_0} (1 + \chi) \theta^2$

$$\chi = \frac{E \alpha_{th}^2 T_0}{\rho_0 C_0}$$

3D $\rho_0 \psi(\theta, \underline{\varepsilon}) = \frac{1}{2} (\lambda \varepsilon_I^2 + 4\mu \varepsilon_{II}) - (3\lambda + 2\mu) \alpha_{th} \theta \varepsilon_I - \frac{\rho_0 C_0 \theta^2}{2T_0}$

□ équations d'état

$$\left| \begin{array}{l} \sigma_{ij}^r = \rho \frac{\partial \psi}{\partial \varepsilon_{ij}} = \lambda \varepsilon_I \delta_{ij} + 2\mu \varepsilon_{ij} - (3\lambda + 2\mu) \alpha_{th} \theta \delta_{ij} \\ s = - \frac{\partial \psi}{\partial \theta} = \frac{1}{\rho} (3\lambda + 2\mu) \alpha_{th} \varepsilon_I + \frac{\rho_0 C_0 \theta}{T_0} \end{array} \right.$$

□ potentiel de dissipation et équations complémentaires

$$\phi(\vec{q}, \underline{\dot{\varepsilon}}) = \frac{k^{-1} \vec{q} \cdot \vec{q}}{2T_0}$$

k tenseur de conduction

$$\left| \begin{array}{l} \underline{\underline{\sigma}}^{ir} = \frac{\partial \phi}{\partial \underline{\dot{\varepsilon}}} = \underline{\underline{0}} \\ \vec{q} = -k \overline{\text{grad}} \theta \quad \text{Fourier} \end{array} \right.$$

$$I_\varepsilon(\varphi) = \int_{\Omega} V(\varphi(x)) dx \quad \lim_{\varepsilon \rightarrow 0^+} I_\varepsilon(\varphi) = W(\cdot, F) = \infty$$

□ Aspects thermiques et calorifiques

Equation générale de diffusion

$$\rho_0 C_0 \dot{T} + \operatorname{div} \mathbf{q} = \underline{\underline{\sigma}}^{\text{ir}} : \dot{\underline{\underline{\epsilon}}} - \mathbf{A} \cdot \dot{\boldsymbol{\alpha}} + T \underline{\underline{\sigma}}^r_{,T} : \dot{\underline{\underline{\epsilon}}} + T \mathbf{A}_{,T} \cdot \dot{\boldsymbol{\alpha}} + r_e$$

Matériau thermoélastique isotrope homogène

$$\rho_0 C_0 \dot{T} - k \Delta T = \cancel{\underline{\underline{\sigma}}^{\text{ir}} : \dot{\underline{\underline{\epsilon}}}} - \cancel{\mathbf{A} \cdot \dot{\boldsymbol{\alpha}}} + T \cancel{\underline{\underline{\sigma}}^r_{,T} : \dot{\underline{\underline{\epsilon}}}} + T \cancel{\mathbf{A}_{,T} \cdot \dot{\boldsymbol{\alpha}}} + r_e$$

Source thermoélastique

$$s_{the} = T \underline{\underline{\sigma}}^r_{,T} : \dot{\underline{\underline{\epsilon}}} = T \frac{\partial^2 \psi}{\partial T \partial \epsilon_{ij}} \dot{\epsilon}_{ij} = -(3\lambda + 2\mu) T \alpha_{th} \operatorname{tr}(\dot{\epsilon}_{ij}) = -3KT \alpha_{th} \dot{\epsilon}_I$$

Modèle simple (i.e. différentiel) de diffusion [AC, Photomécanique, 95]

$\theta = T - T_0$ (HPPT $\theta \ll T_0$), sources homogènes, fuites linéaires, ...

la solution spectrale limitée au 1^{er} vecteur propre...



$$A_i = \partial \psi / \partial \alpha_i$$

Variables internes

$$\dot{\theta} + \frac{\theta}{\tau_{th}} = - \frac{3KT\alpha_{th}}{\rho_0 C_0} \dot{\epsilon}_I$$

□ Application à l'analyse thermique des contraintes (TSA)

P. Stanley & J. Barton (UK), J. Rowlands (AUS), U. Galiotti (IT), ...
Dispositif SPATE (79), Deltatherm (96) ...

$$s_{the} = -3KT\alpha_{th}\dot{\varepsilon}_I = -T\alpha_{th}\dot{\sigma}_I - 9K\alpha_{th}^2\dot{\theta}$$

$$\left(1 + \frac{9K\alpha^2 T_0}{\rho_0 C_0}\right)\dot{\theta} - \frac{k}{\rho_0 C_0} \Delta\theta = -\frac{\alpha_{th} T}{\rho_0 C_0} \dot{\sigma}_I$$

$\underbrace{\qquad\qquad\qquad}_{<< 1}$

diffusivité

Si processus adiabatique et faibles variations de température ...

$$\delta\theta = -\frac{T_0\alpha_{th}}{\rho_0 C_0} \delta\sigma_I = K_{TSA} \delta\sigma_I$$

« Vous mesurez des températures, vous mesurez des contraintes ! »

TSA : un exemple d'application industrielle



New developments in Thermo Elastic Stress Analysis by Infrared Thermography.

P. Bremond – Cedip IR systems

IV Conferencia Panamericana de END Buenos Aires - 2007

Pierre Bremond
Agema – Additionnal
Cedip – Flir

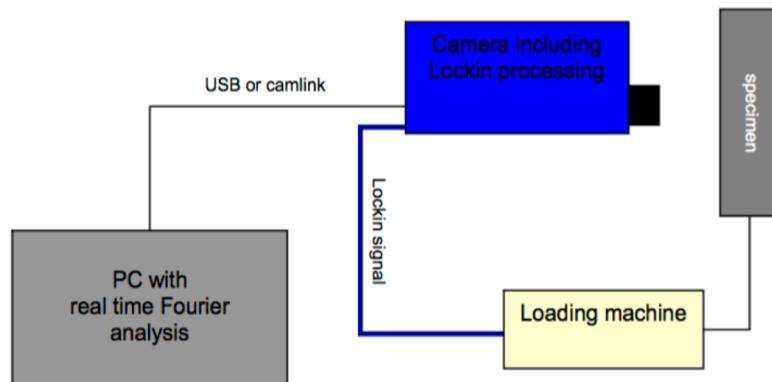
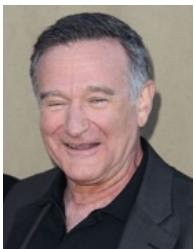


Figure 1. Signal processing into Thermographic Stress Analyzer



SFT
GDR 2519
PM
SEM

R. Williams

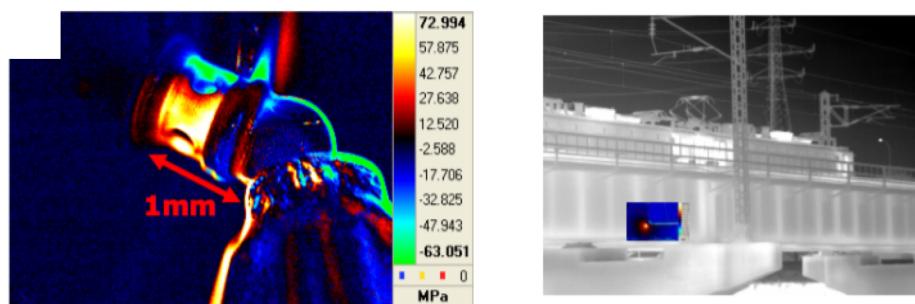


Figure 4. Size of object analyzed by TheSA

TheSA systems have a large panel of applications. Basically the use of imaging measurement systems is required when localization of high stress is unknown or to compare to another full field data like FEA results.

$$I_\varepsilon(\varphi) = \int_{\Omega} V(\varphi(\boldsymbol{x})) \, d\boldsymbol{x} \quad \lim_{\varepsilon \rightarrow 0^+} I_\varepsilon(\varphi) = W(\cdot, F) = \infty$$

TSA : bilan qualitatif

Points forts :

- Méthode « simple » à mettre en œuvre
- Champ du premier invariant du tenseur des contraintes
- Contraintes planes :
 - TSA + Photoélasticimétrie → contraintes principales
 - Cf. [Barone, Wang, Patterson, IUTAM 1998]
- Concentration de contraintes sur structures planes, biréfringentes

Limitations :

- Mesure de θ : pièces massives complexes, corps lambertiens,
- Diffusion, rayonnement parasites... stabilité thermique...
- Adiabaticité
- Comportement élastique ... caractéristiques homogènes, constantes ...

$$\delta\theta = -\frac{T_0 \alpha_{th}}{\rho_0 C_0} \delta\sigma_I$$

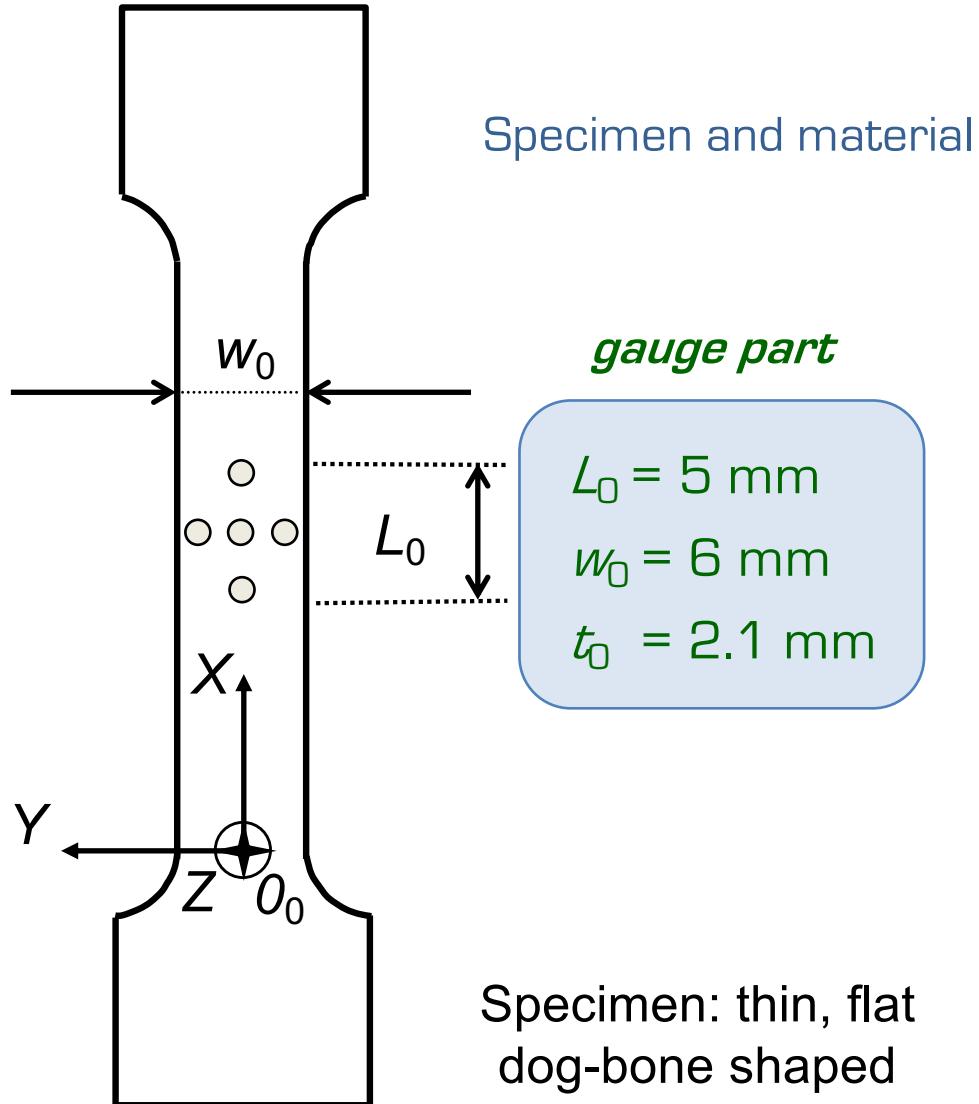
$$I_\varepsilon(\varphi) = \int_{\mathbb{R}^n} V(\varphi(x)) dx + \frac{1}{2\varepsilon} \int_{\mathbb{R}^n} |\nabla \varphi(x)|^2 dx$$



Elasticité caoutchoutique
Inversion thermoélastique



Rubber elasticity [Caborgan et al., 2009]



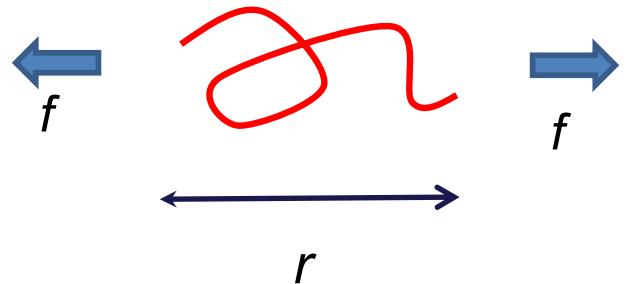
« natural » rubber (NR)

cis-1,4 polyisoprene,
per 100 g 3 g stearic acid,
9.85 g zinc oxide,
2 g antioxidant,
3 g sulphur,
3 g plasticiser,
4 g accelerators

heated at 160 °C for 10 minutes,
stored at - 10 °C, in dry air

Rubber elasticity: molecular approach (I)

Molecular chain



$$f = \frac{3kT}{\langle r_0^2 \rangle} r$$

Entropic elasticity

$$f = \left(\frac{\partial \Psi}{\partial r} \right)_T = \cancel{\left(\frac{\partial E}{\partial r} \right)_T} - T \left(\frac{\partial S}{\partial r} \right)_T$$

$$S = k \log(\Omega)$$

Ω : number of conformations

k : Boltzmann constant

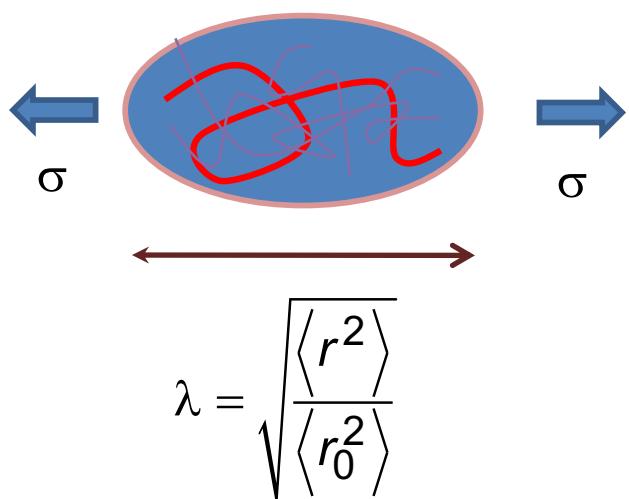
$\langle r_0^2 \rangle$ m. s. end-to-end distance of the chain

- f vs. r : linear \Rightarrow « elasticity »
- f vs. T : $F \nearrow$ with T !! \Rightarrow different from standard thermoelasticity where $F \searrow$ with T + volume variation (i.e. thermodilatability) !!

Rubber elasticity: molecular approach (II)

Chain → Network (N_{ch} chains)

Statistical modeling of molecular network



- incompressibility assumption
- uniaxial loading in the direction 1

- affine model [Flory, 1953]
- isotropic material

$$\Delta \psi_{\text{nwk}} = -T \Delta S_{\text{nwk}} = \frac{N_{\text{ch}} k T}{2} (I_1 - 3)$$

$$I_1^2 = \text{tr}(F^T F) = \sum_{i=1}^{i=3} \lambda_i^2$$

$$F = \frac{\partial x}{\partial X} \quad \begin{array}{l} \text{transformation} \\ \text{gradient tensor} \end{array}$$

$$\lambda_i \quad \begin{array}{l} \text{extension ratios} \end{array}$$

tensile Cauchy stress

$$\sigma_{11} = N_{\text{ch}} k T (\lambda_1^2 - \lambda_1^{-1}) = N_{\text{ch}} k T (e^{2\varepsilon_{11}} - e^{-\varepsilon_{11}})$$

with ε the Hencky strain tensor

$$I_\varepsilon(\varphi) = \int_{\Omega} V(\varphi(\boldsymbol{x})) \, d\boldsymbol{x} \quad \lim_{\|\boldsymbol{F}\|_F \rightarrow 0^+} W(\cdot, \boldsymbol{F}) = \infty$$

Rubber elasticity: mechanical approach (III)

Hyperelasticity : continuum thermomechanical approach

- non linear elasticity at finite strain
- objectivity (strain invariants)
- initial isotropy
- isochoric transformation
- often isothermal models

A lot of « phenomenological » models, e.g. Mooney-Rivlin (1948), [Ogden, ...]

$$\boxed{\psi(\lambda_i) = C_1(I_1 - 3) + C_2(I_2 - 3)} \xrightarrow{\text{tension}} \boxed{\sigma_{11} = 2(\lambda_1^2 - \lambda_1^{-1})(C_1 + C_2\lambda_1^{-1})}$$

$$\text{with } I_1 = \lambda_1^2 + \lambda_2^2 + \lambda_3^2 \quad I_2 = \lambda_1^2\lambda_2^2 + \lambda_2^2\lambda_3^2 + \lambda_3^2\lambda_1^2 \quad I_3 = \lambda_1^2\lambda_2^2\lambda_3^2 = 1$$

$$I_\varepsilon(\varphi) = \int_{\Omega} V(\varphi(x)) \, dx \quad \lim_{\varepsilon \rightarrow 0^+} I_\varepsilon(\varphi) = W(\cdot, F) = \infty$$

Rubber elasticity: modified entropic elasticity [III]

(i) Entropic effects : Gough (1805) - Joule (1857)

$$e(s, \varepsilon) = e_c(T) \quad \text{analogy of perfect gaz} \quad \psi_c(T, \varepsilon) = T K_1(\varepsilon) + K_2(T)$$

(ii) Inversion of thermoelastic effect : Anthony (1942)

standard thermoelasticity at low extension ratio, rubber effects at high strain ratio

[i], [ii] literature: modified entropic elasticity

$$e(s, \varepsilon) = e_1(T) + e_2(\varepsilon) \quad [\text{Chadwick and Creasy, 1984}]$$

drawback : strongly non linear expression,
many parameters hardly identifiable

$$I_\varepsilon(\varphi) = \int_{\Omega} V_\varepsilon(\varphi) \, dx \quad \lim_{\varepsilon \rightarrow 0^+} I_\varepsilon(\varphi) = W(\cdot, F) = \infty$$

Rubber behavior: a simple heuristic model (IV)

Proposal:

- « leave » the classical framework of (entropic) elasticity
- interpret the « thermoelastic inversion » as a competition between 2 coupling mechanisms
- describe this competition introducing a « rubber strain tensor » ε_c in a series model

$$\psi(T, \varepsilon, \varepsilon_c) = \psi_{\text{the}}(T, \varepsilon - \varepsilon_c) + \psi_c(T, \varepsilon_c)$$

standard thermoelastic free energy free energy of affine model

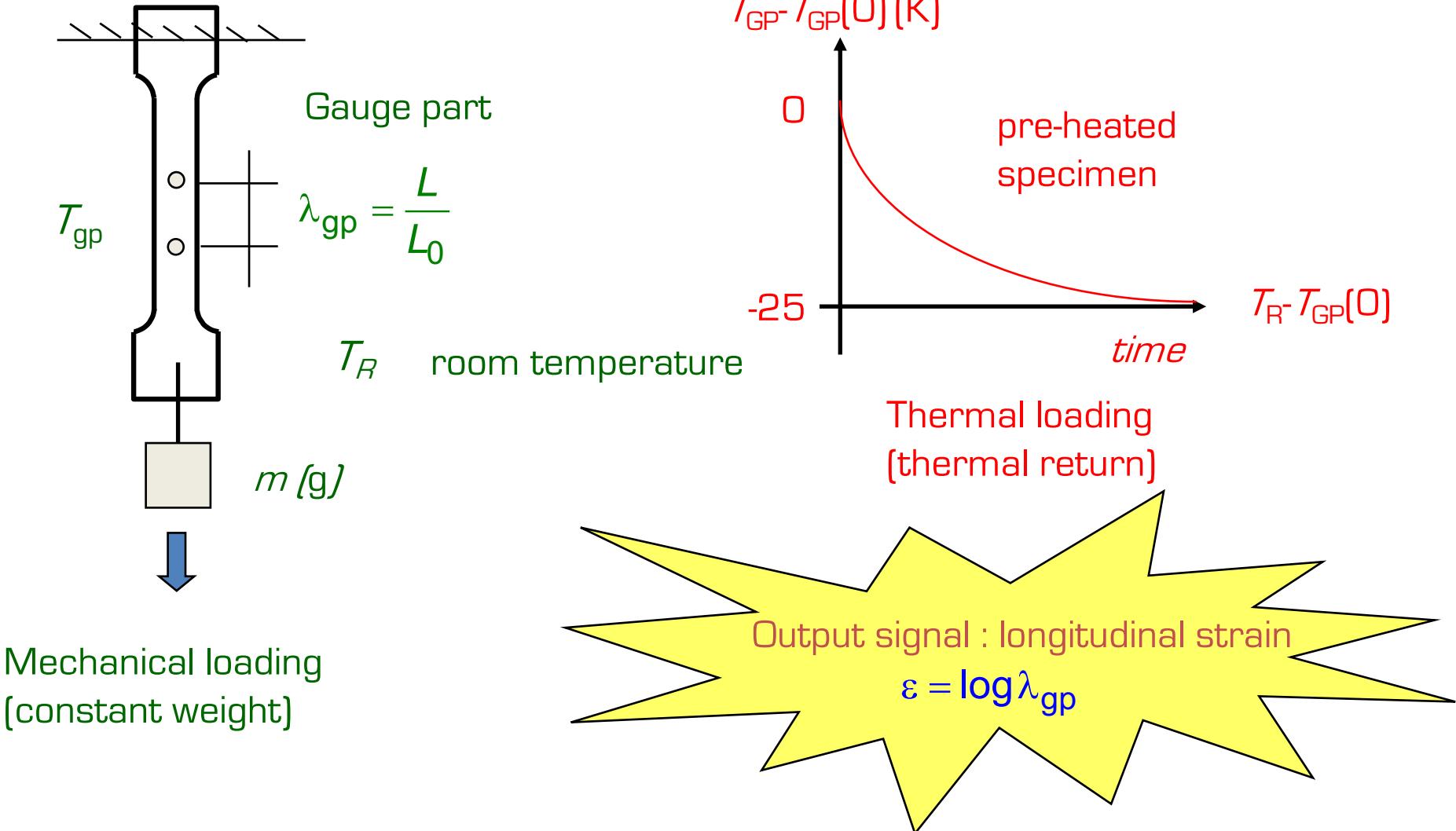


a first step ... towards possible irreversibility ; Mullins effects [Mullins, 1948]

$$\varphi(q, \dot{\varepsilon}, \dot{\varepsilon}_c; T) = \frac{q^2}{2kT} + \eta \dot{\varepsilon}_c^2$$



Thermoelastic inversion : protocol



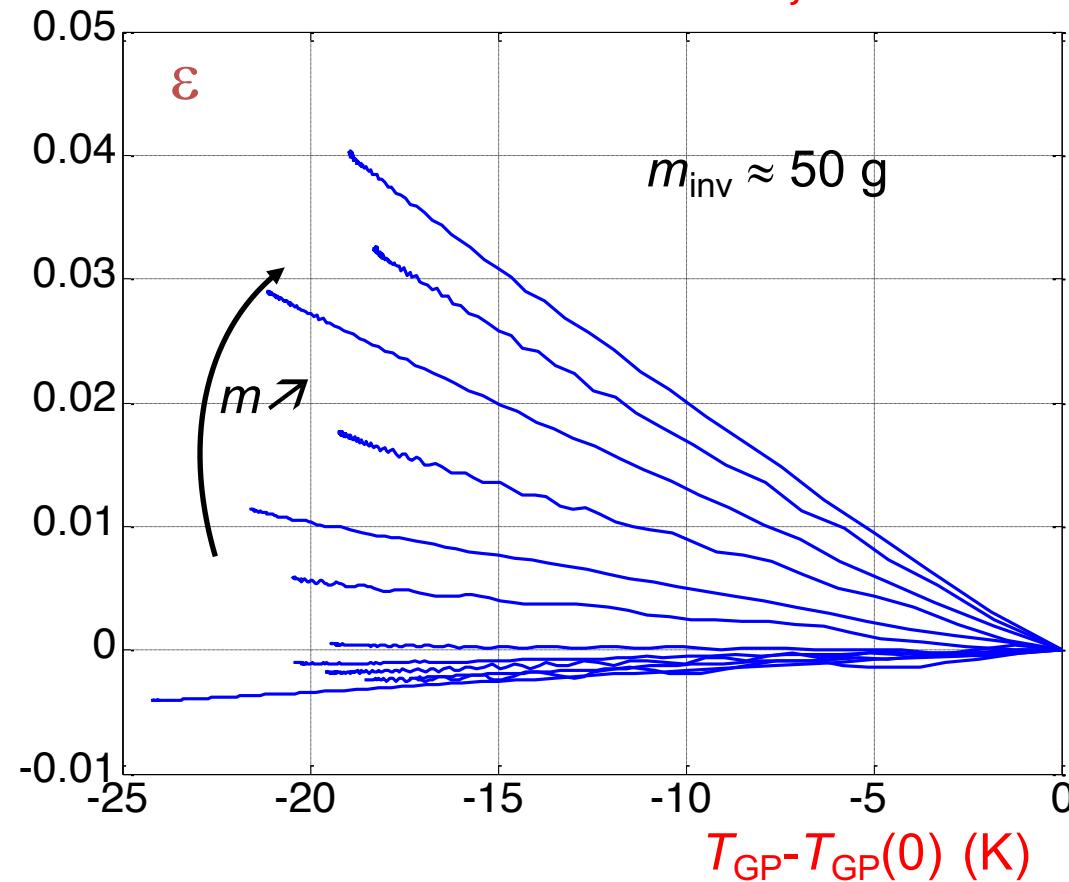
Thermoelastic inversion : results

Anthony , 1942: relaxation test



PHD Caborgan 2010 : creep test

No viscosity here !?



$m = 0 \quad 3 \quad 30 \quad 40 \quad 50 \quad 60 \quad 100 \quad 200 \quad 300 \quad 400 \quad 520 \quad 640 \quad 730$

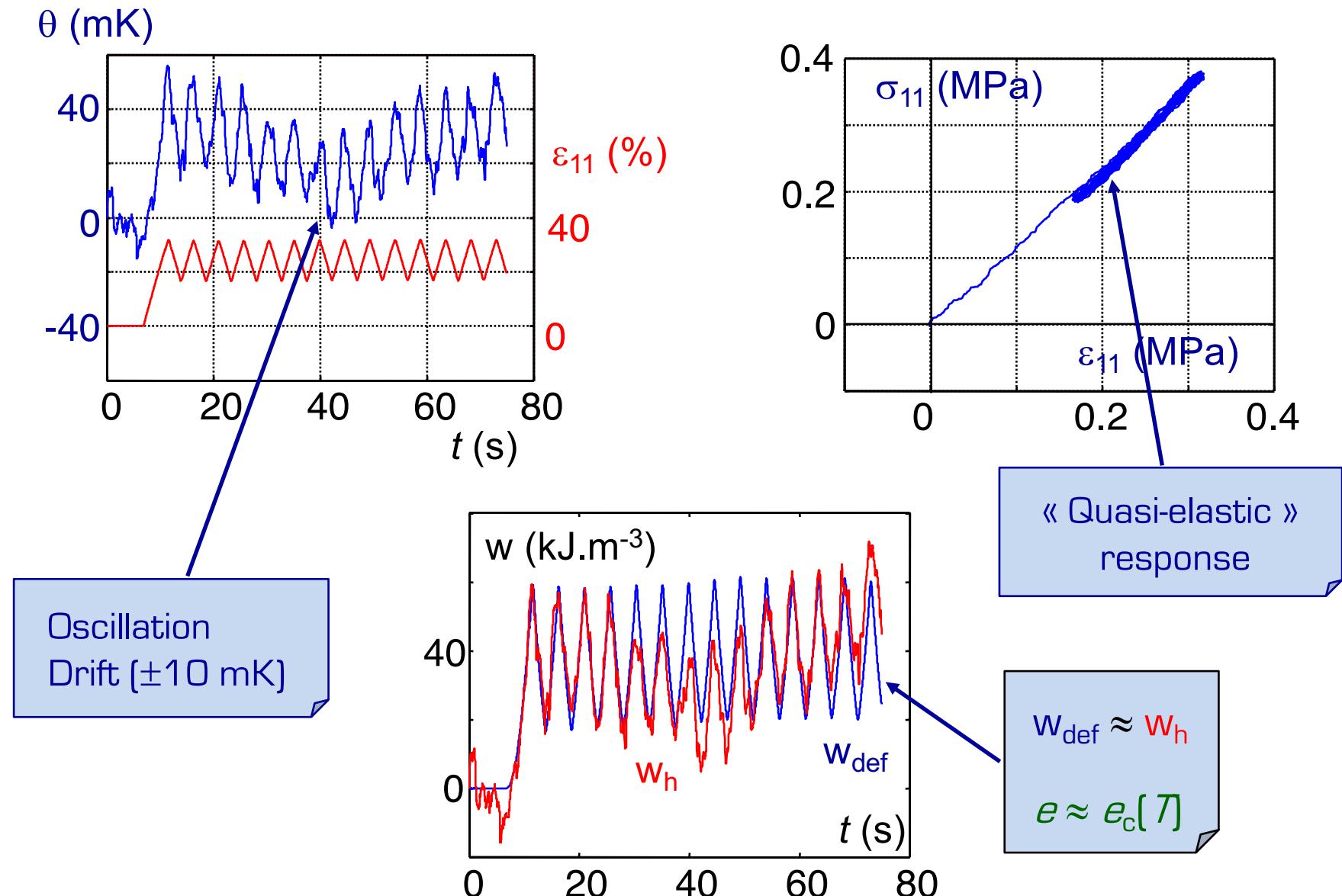
$\varepsilon \leq 0 \text{ if } m \leq m_{\text{inv}}$

Predominance of
thermoelastic coupling

$\varepsilon > 0 \text{ if } m > m_{\text{inv}}$

Predominance of
rubber coupling

Load-unload cycles : experimental results



$$I_\varepsilon(\varphi) = \int_{\Omega} V(\varphi(x)) \, dx + \lim_{\delta \rightarrow 0^+} \frac{1}{\delta} \int_{\Omega} F_\delta(\nabla \varphi) \, dx$$

Euristic 1D model: identification

ψ_c

volume free energy

$$\psi(\theta, \varepsilon, \varepsilon_c) = \underbrace{\frac{E}{2} (\varepsilon - \varepsilon_c - \alpha_{th}\theta)^2}_{\psi_{the}} - \left(\frac{\rho C}{T_0} + E\alpha_{th}^2 \right) \frac{\theta^2}{2} + K_c(T_0 + \theta) \left(\frac{e^{2\varepsilon_c}}{2} + e^{-\varepsilon_c} - \frac{3}{2} \right)$$

dissipation potential

$$\varphi(q, \varepsilon, \varepsilon_c) = \frac{q^2}{2kT_0} + \eta \dot{\varepsilon}_c^2$$

material constants

$\rho = 950 \text{ kg.m}^{-3}$, $C = 1150 \text{ J.kg}^{-1}.K^{-1}$, $\tau_{eq} = 20 \text{ s}$ (OD diffusion model)

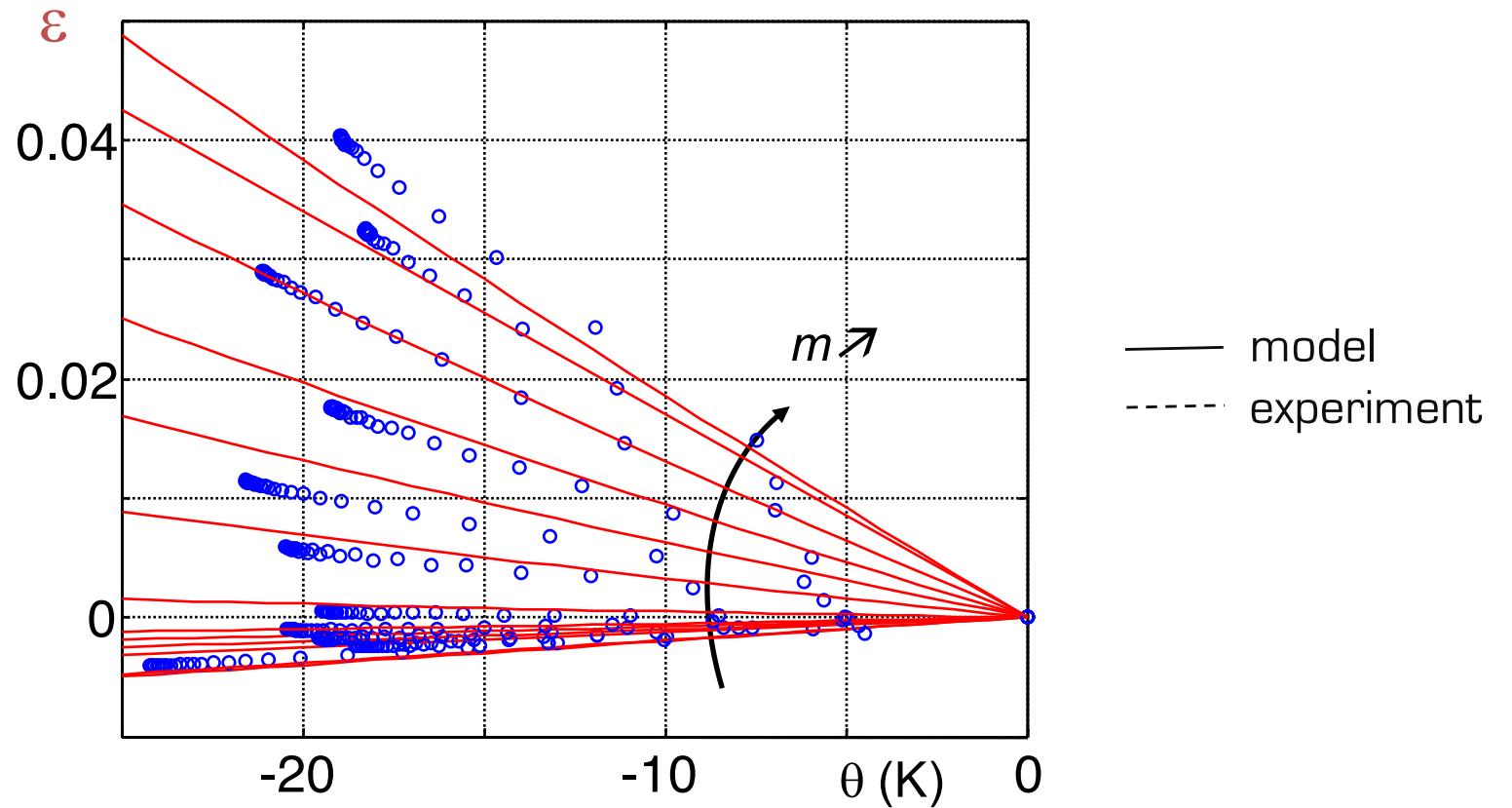
$E = 100 \text{ MPa}$, $\alpha_{th} = 2.10^4 \text{ K}^{-1}$

$K_c = 1110 \text{ Pa.K}^{-1}$

$\eta = 0 \text{ MPa.s}$, no viscous effect...

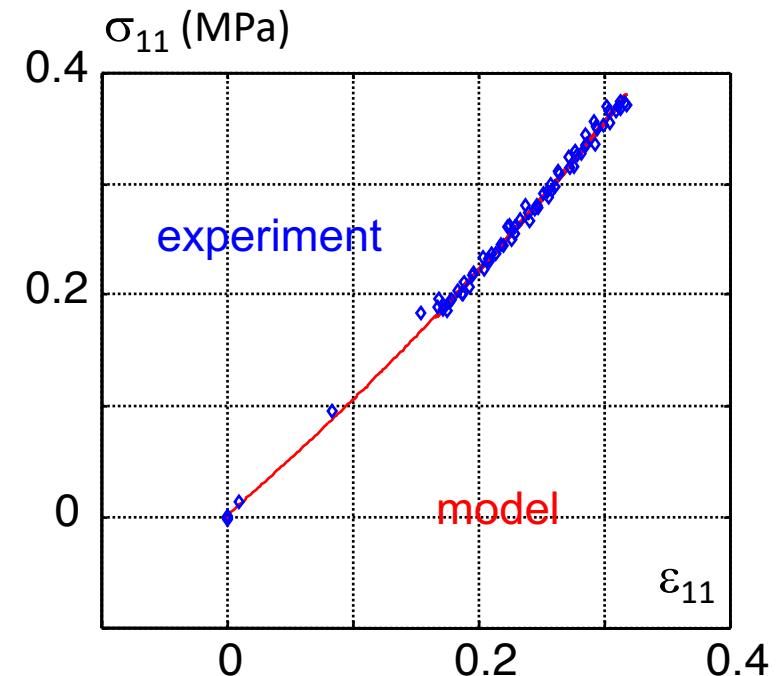
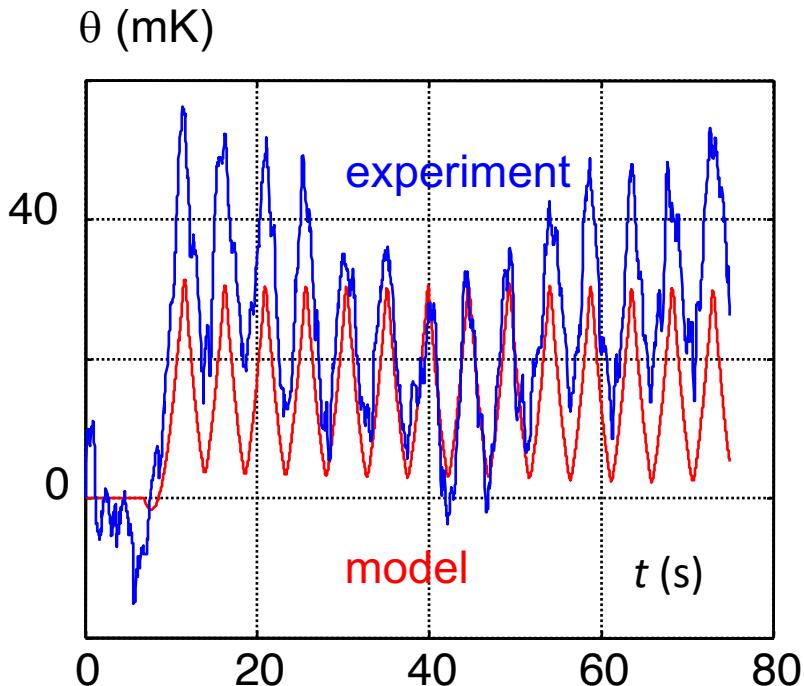
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Thermoelastic inversion : exp. vs. num. responses

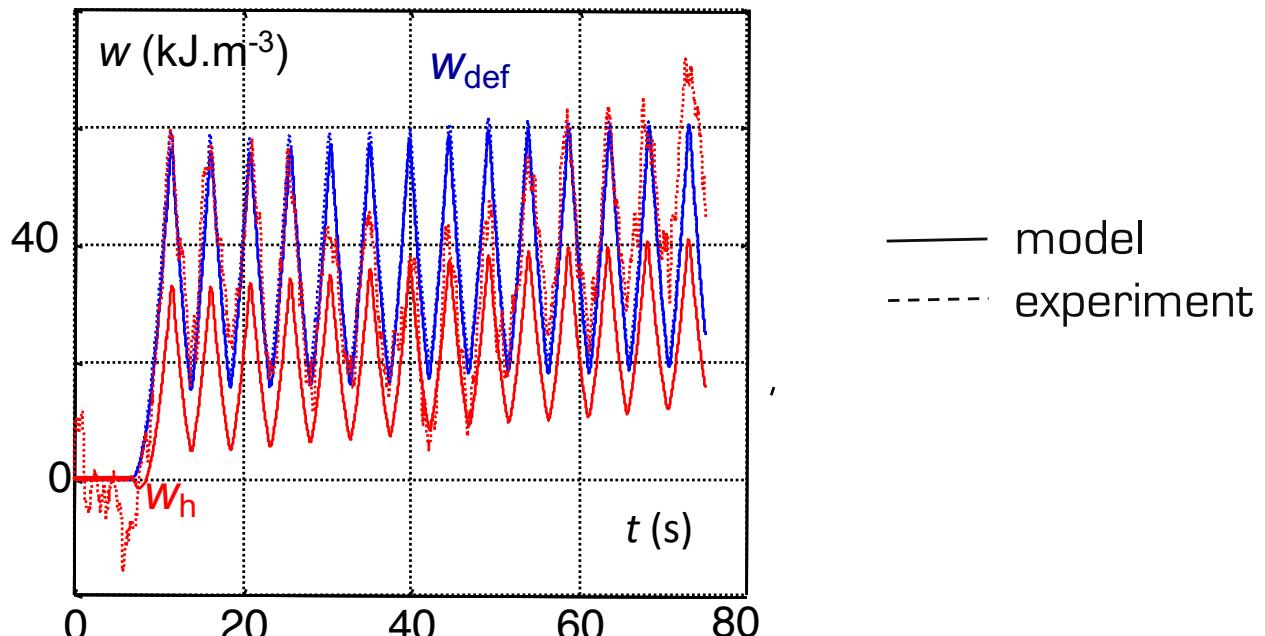


$m = 0 \quad 3 \quad 30 \quad 40 \quad 50 \quad 520 \quad 640 \quad 730$

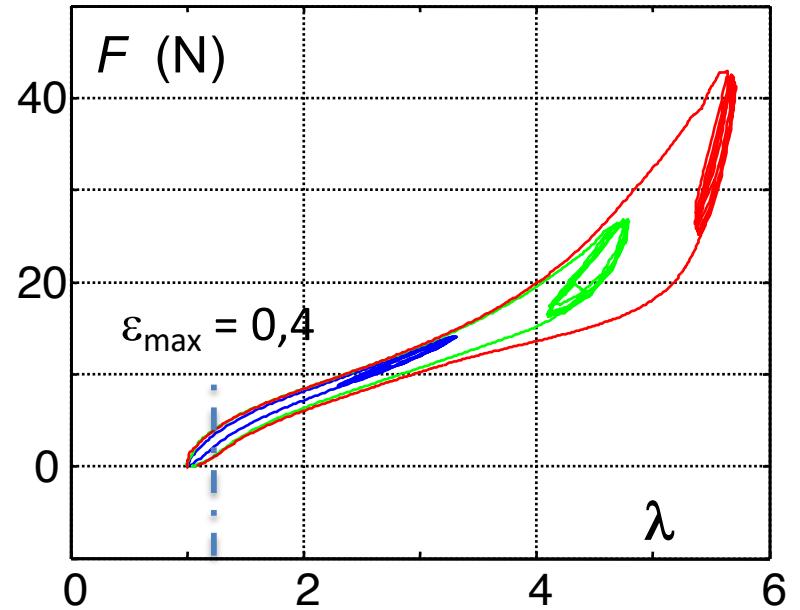
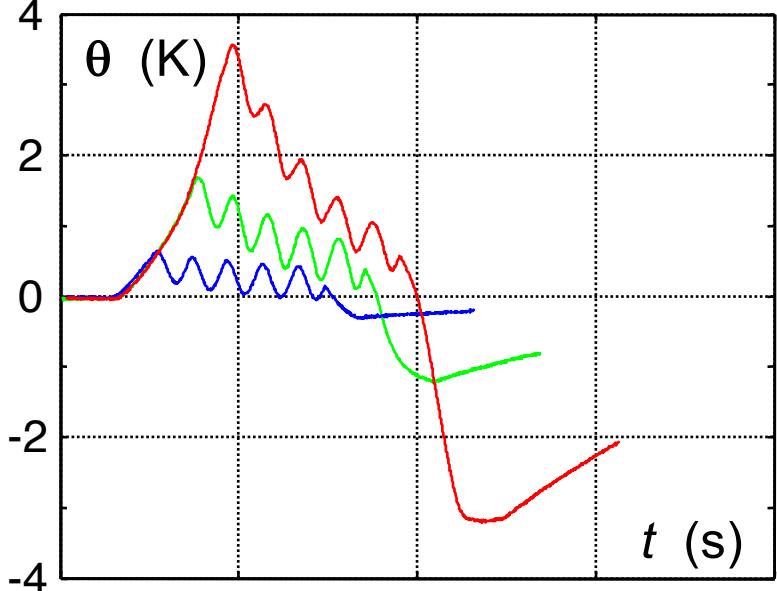
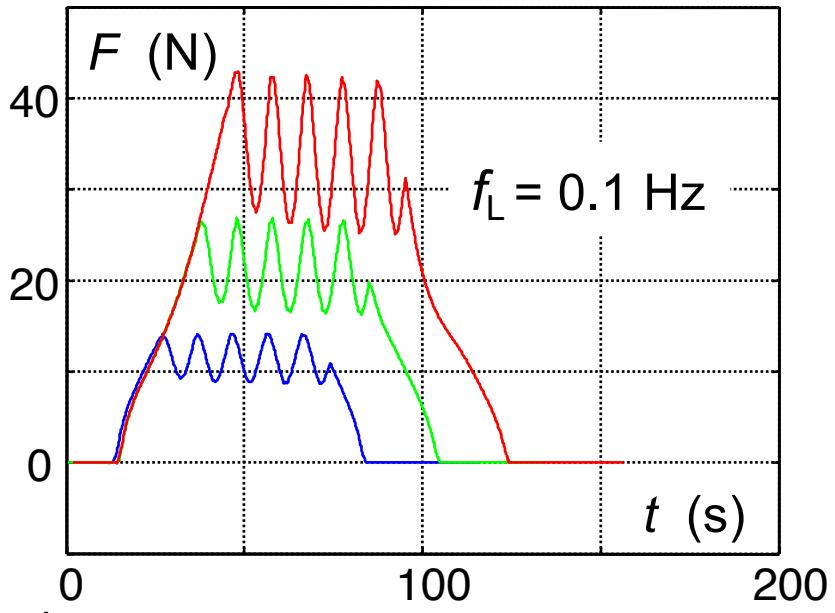
Load-unload cycles : exp. vs. num. responses



$\epsilon_{\max} = 0,4$



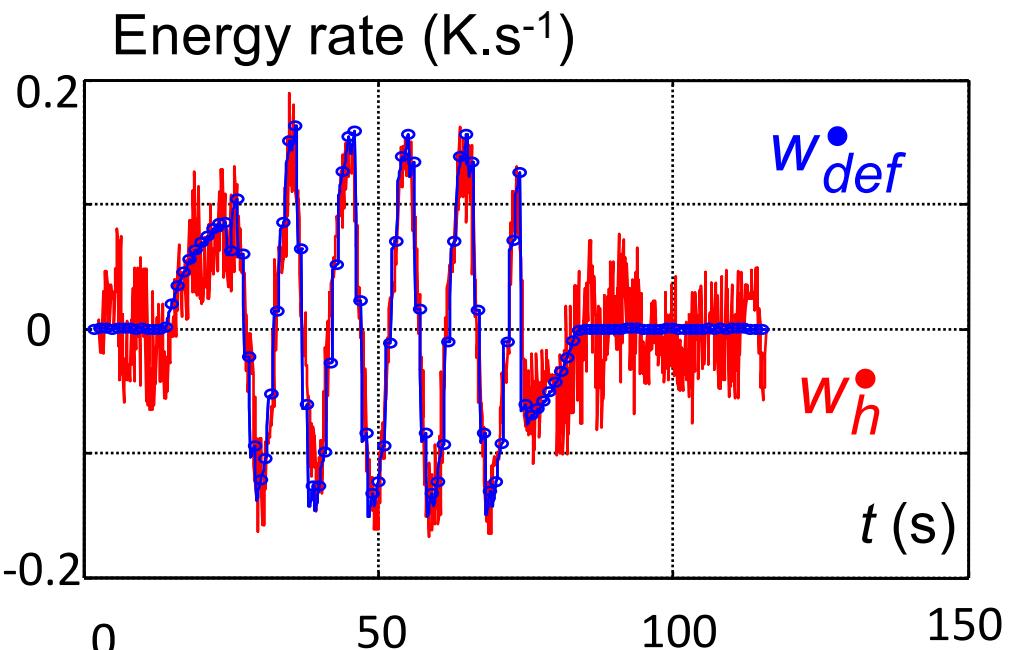
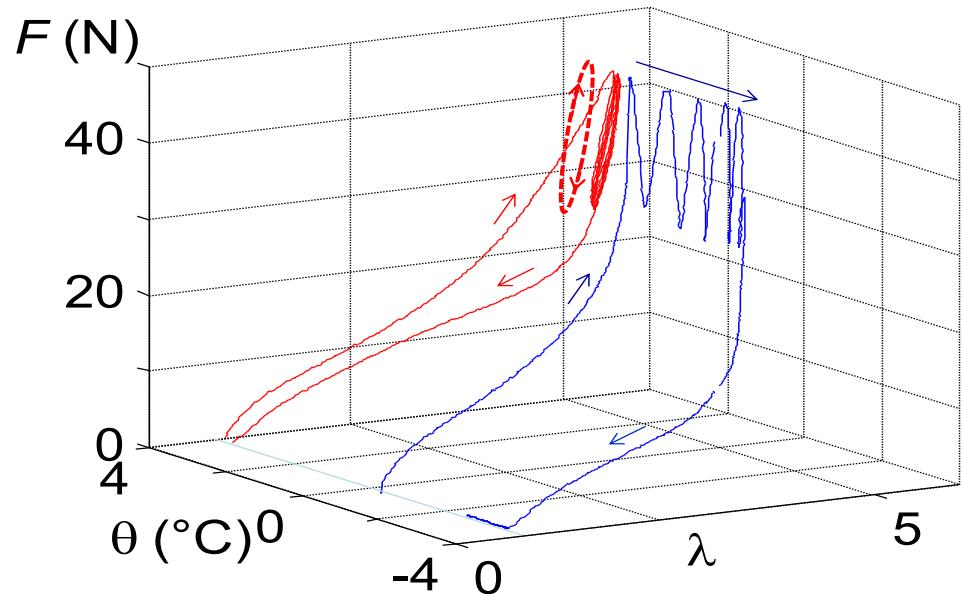
Load-unload test (large strain)



- 3 tests , 5 « internal » loops
- displacement controlled tests
- hysteresis loops
- thm couplings vs. dissipation

Energy balance at large strain

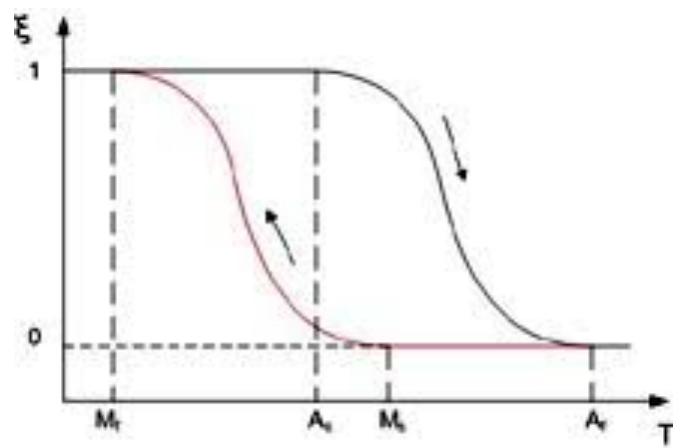
[Caborgan, PhD 11]



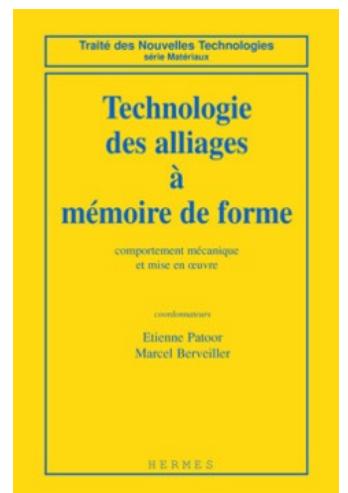
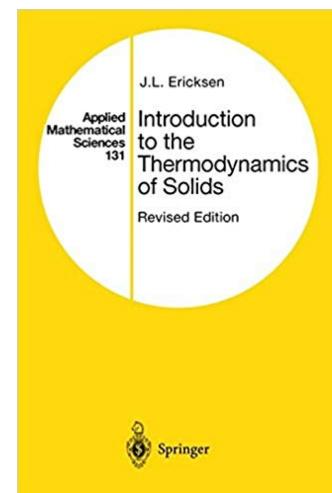
In agreement with the « perfect gaz » hypothesis

$$w_{def}^{\bullet} = w_h^{\bullet}$$

$$I_\varepsilon(\varphi) = \int_{\Omega} V_\varepsilon(\varphi) \, dx \quad \text{and} \quad \lim_{\varepsilon \rightarrow 0^+} I_\varepsilon(\varphi) = W(\cdot, F) = \infty$$

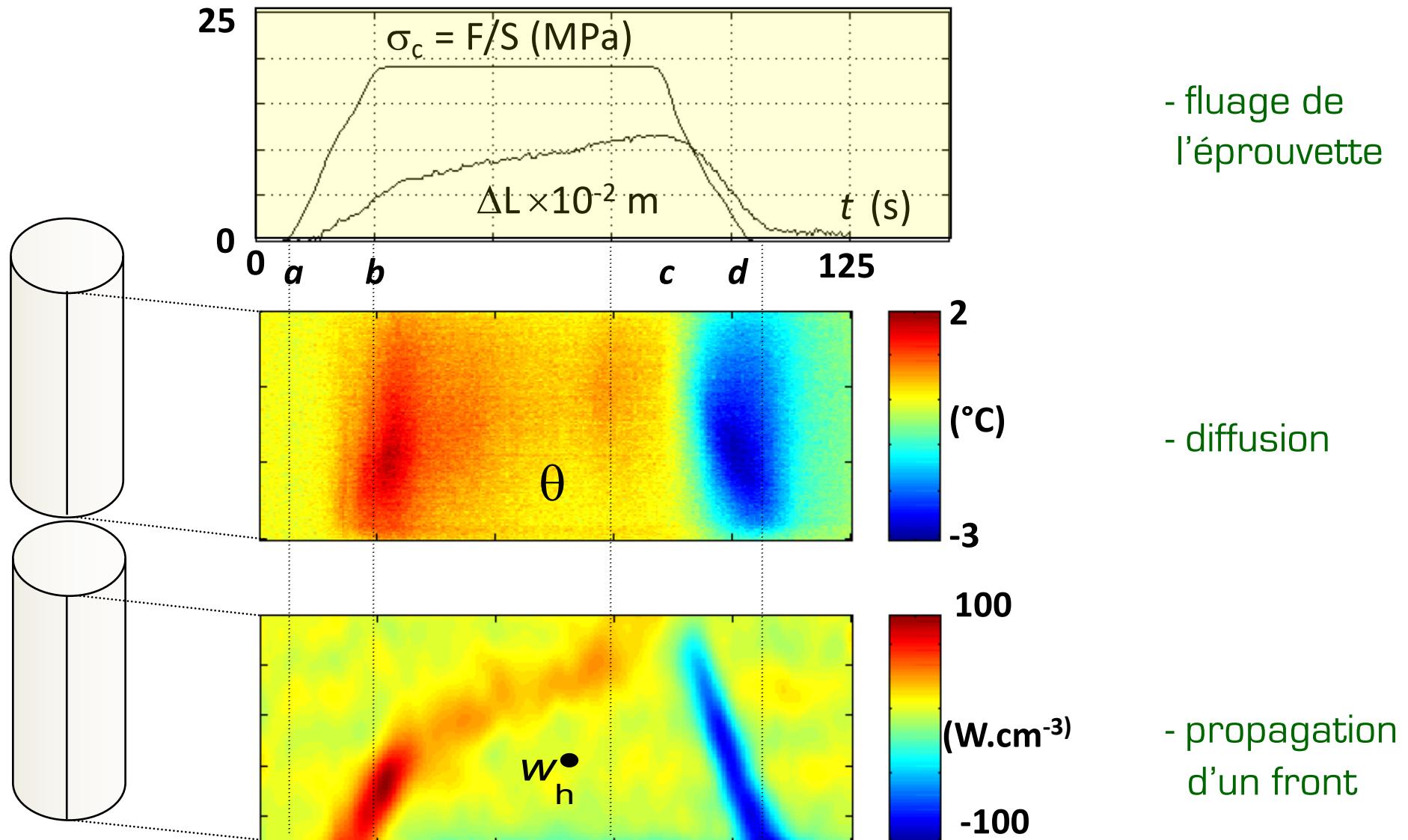


Transition de phase solide-solide Pseudoélasticité des AMF



Pseudoélasticité des AMF : un «vieil» exemple

M. Löbel (95) → X. Balandraud (00) → S. Leclercq (01)
Monocristaux d'AMF CuZnAl CuAlBe



$$I_\varepsilon(\varphi) = \int_{\Omega} V(\varphi(\boldsymbol{x})) \, d\boldsymbol{x} \quad \lim_{\varepsilon \rightarrow 0^+} I_\varepsilon(\varphi) = W(\cdot, F) = \infty$$

Energy balance for “pseudoelastic” SMA

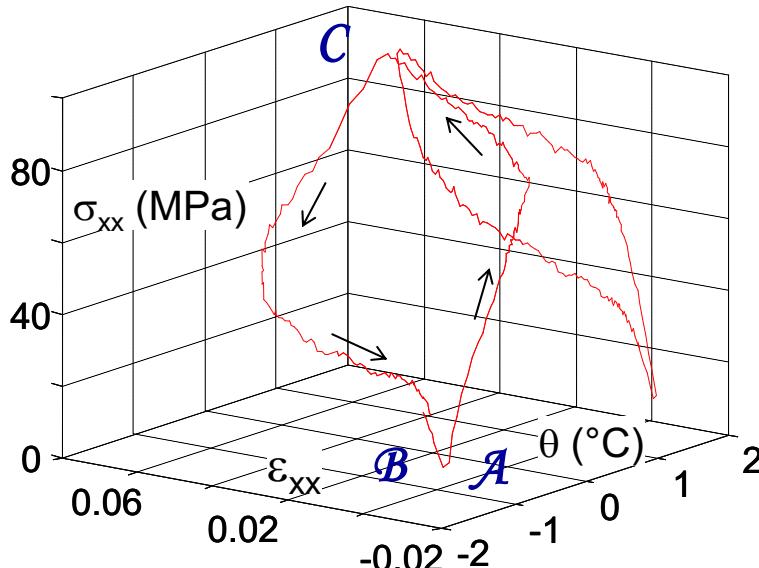
Load-unload cycle

$$\mathcal{A}_h \approx 1,9 \text{ MJ.m}^{-3}$$

$$w_{\text{def}} \Big|_{\mathcal{A}}^C = 4,9 \text{ MJ.m}^{-3}$$

$$w_h \Big|_{\mathcal{A}}^C = 28 \text{ MJ.m}^{-3}$$

CuAlBe



specimen response (macroscale)

$$R_M = \frac{\mathcal{A}_h}{w_{\text{def}}} \approx 0,39$$

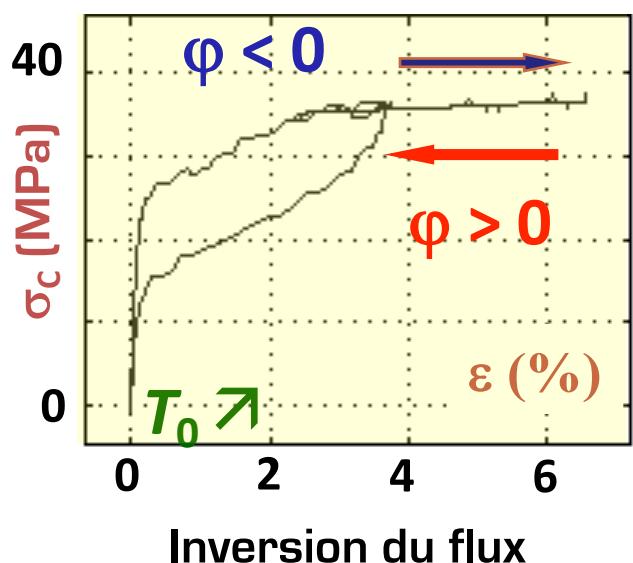
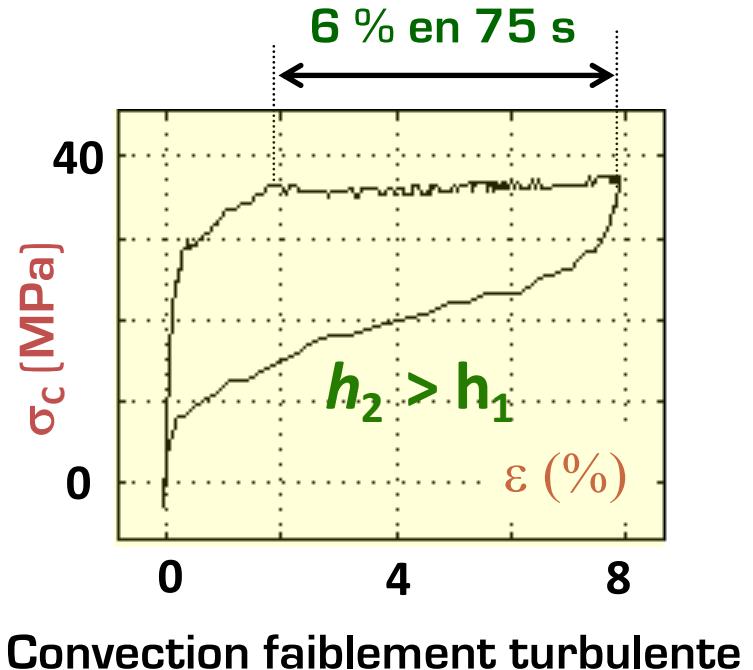
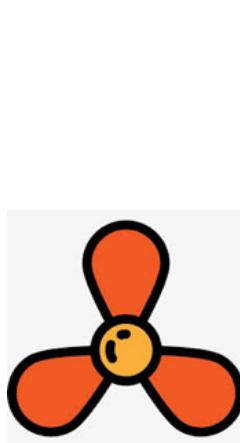
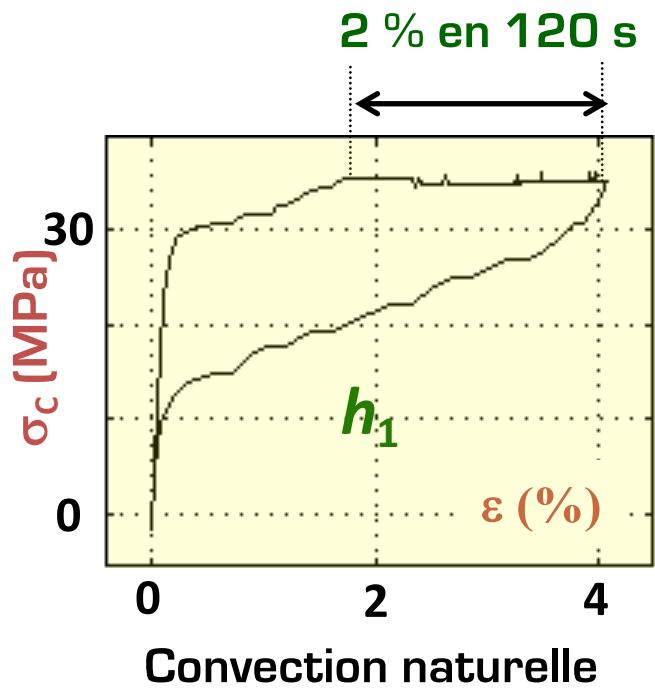
$$0 \leq R_T = \frac{\int w_h \, d\tau}{\int |w_h| \, d\tau} \approx 0,026 \leq 1$$

[S. Vigneron, PhD 09]

$R_T = 0$ non dissipative (intrinsically) + isothermal or adiabatic process

$R_T = 1$ purely dissipative (intrinsically) process without any coupling effect

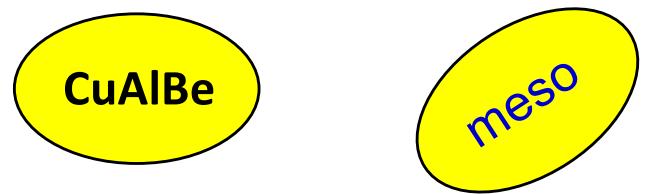
strong thermomechanical couplings



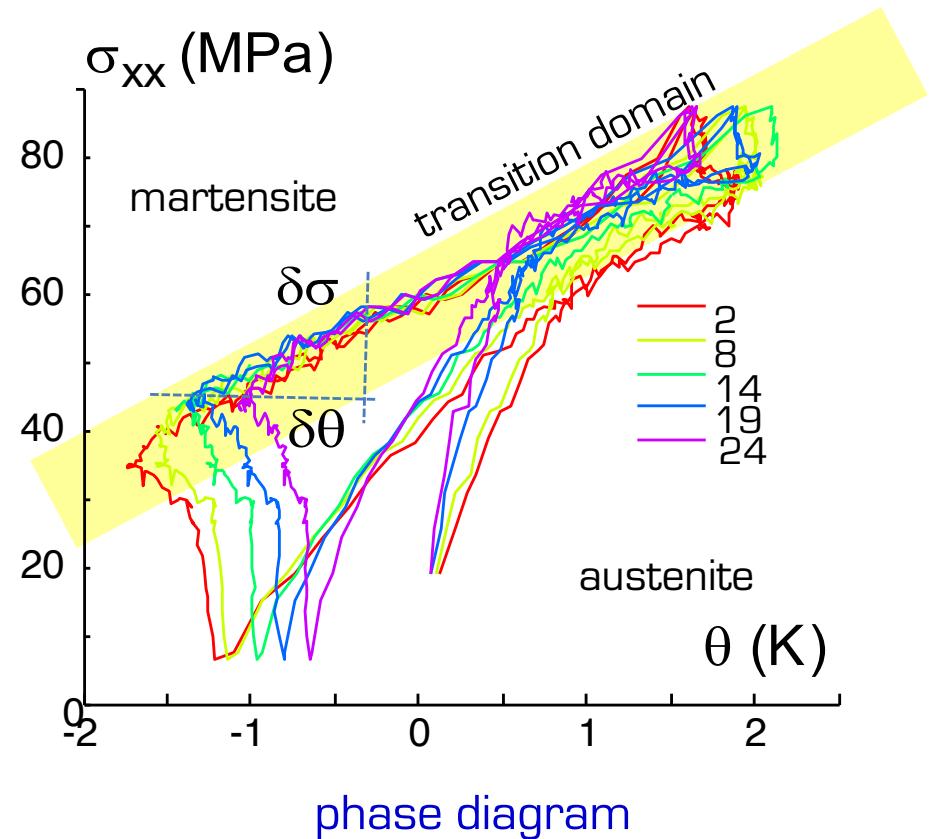
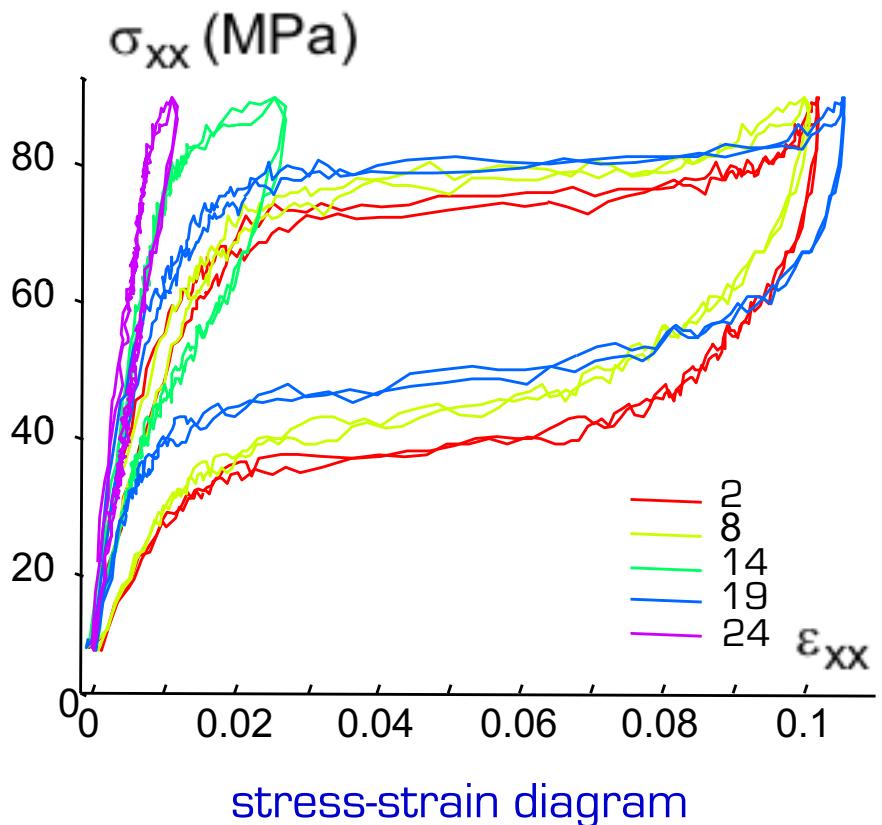
Vitesse de fluage : CL thermiques
Fluage «inverse»
Hystérésis ($R_T < 5\%$)
Comportement ThM couplé :

Transformation de phase
anisotherme sous contrainte

Heterogeneous mesoscopic responses



*Localization of phase change
Propagation of phase change front*



$\delta\theta$ transition domain «thickness» $\sim 1\text{K}$ (15 K for the SMA producer !)

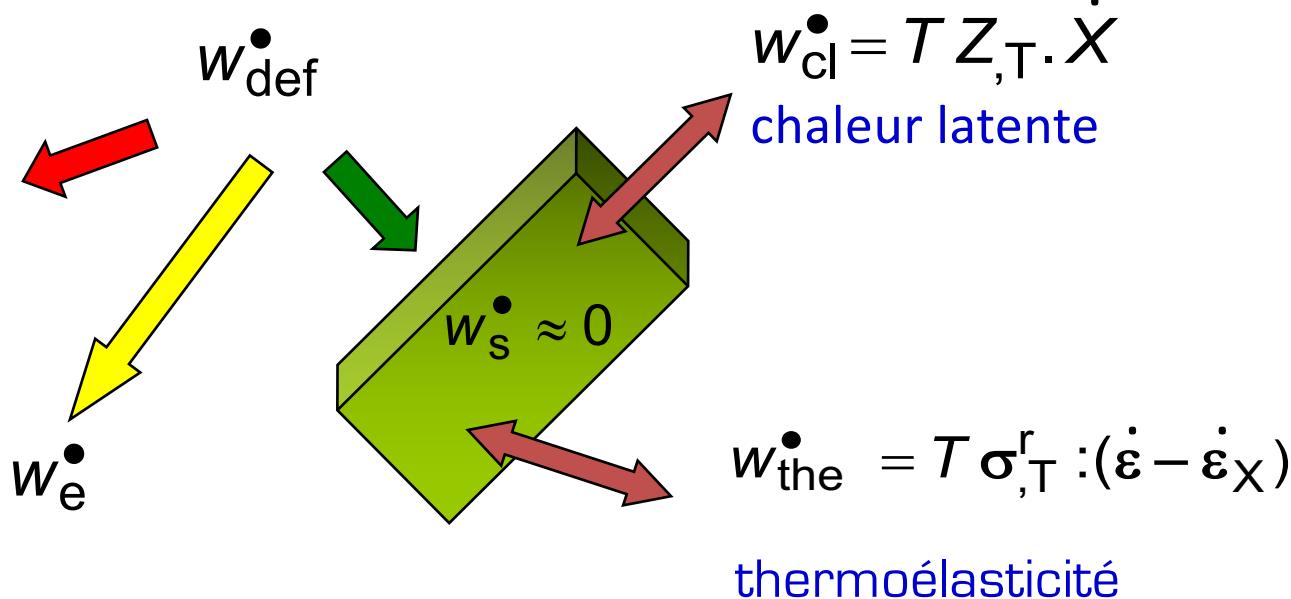
$$I_\varepsilon(\varphi) = \int_{\Omega} V_\varepsilon(\varphi) \, dx \quad \lim_{\varepsilon \rightarrow 0^+} I_\varepsilon(\varphi) = W(\cdot, F) = \infty$$

Modélisation : 1^{ers} ingrédients

- Petites perturbations thm $\theta = T - T_0 \quad T_0$
- Cinématique $\varepsilon = \varepsilon_e + \varepsilon_\theta + \varepsilon_X \quad \varepsilon_X(X) = \sum_n X_n \beta_n$
- Énergie libre $\psi(T, \varepsilon, X) = \psi_{\text{the}}(T, \varepsilon - \varepsilon_X(X)) + \psi_{\text{tr}}(T, X)$
- Potentiel de dissipation $\varphi(\nabla T, \dot{\varepsilon}, \dot{X}) \Rightarrow \varphi(\nabla T)$
- Bilan d'énergie

$$\varphi(\nabla T, \dot{\varepsilon}, \dot{X}) \Rightarrow \varphi(\nabla T)$$

$$\left| \begin{array}{l} \sigma^{\text{ir}} = 0 \\ Z = 0 \\ \text{si } \dot{X} \neq 0 \end{array} \right.$$



$$I_\varepsilon(\varphi) = \int_{\Omega} V(\varphi(x)) dx \quad \lim_{\varepsilon \rightarrow 0^+} I_\varepsilon(\varphi) = W(\cdot, F) = \infty$$

Modèle ThM-1D

- traction

$$\sigma \geq 0$$

- quasi-staticité

$$\sigma_x = 0$$

- cinématique

$$\varepsilon = \frac{\sigma}{E} + \alpha \theta + \beta X$$

- cinétique [austénite-martensite]

$$\dot{Z}X = 0$$

force nulle si
transformation de phase

Constantes

E : Module d'Young

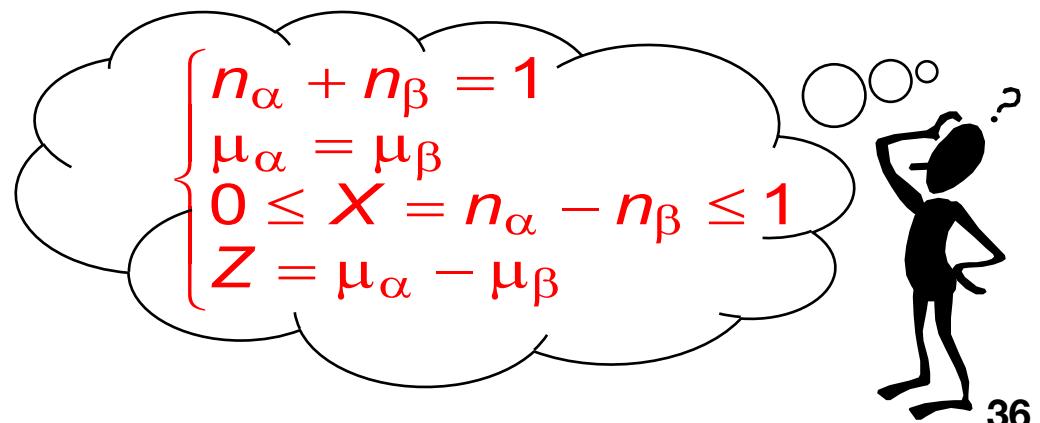
α : dilatation

β : transformation

k : conduction

ρ : masse volumique

C : chaleur spécifique



Modèle ThM-1D

- diagramme de phase

«Martensite» : G.B. Olson & W.S. Owen
Chap. 12, 1991.



- Cinématique [algébrique !]

$$X = \min\left(1, \left\langle \frac{\sigma - K(T - A)}{K(A - M)} \right\rangle^+\right)$$

- existence d'un front

$$A - M \ll A \quad A - M \approx 10^{-3} {}^\circ C$$

i.e. localisation des déformations et des sources de chaleur

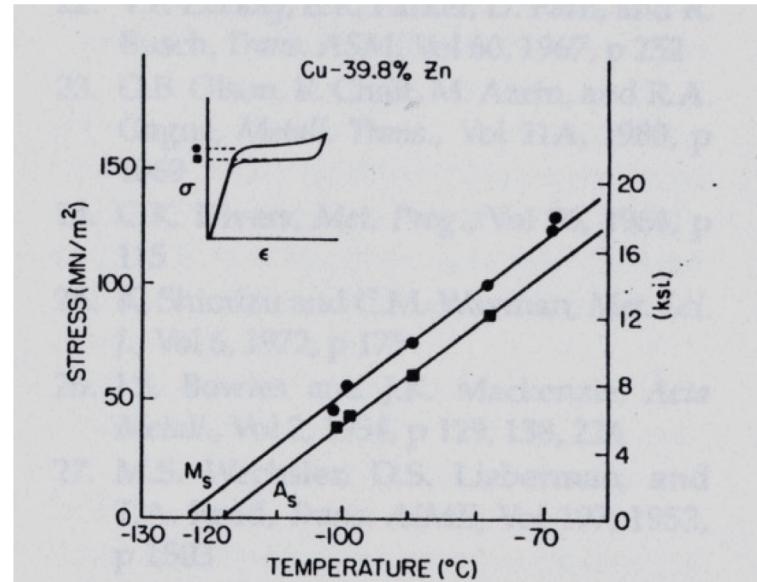
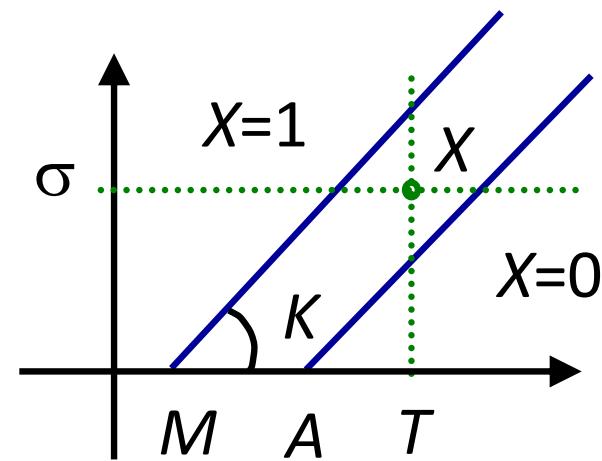


Fig. 9 Temperature dependence of stress required to produce stress-induced martensite in a Cu-Zn alloy

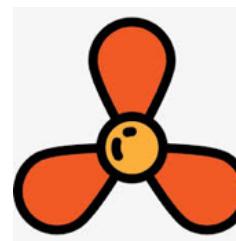


Essai numérique (I)

Coefficient d'échange et vitesse du front

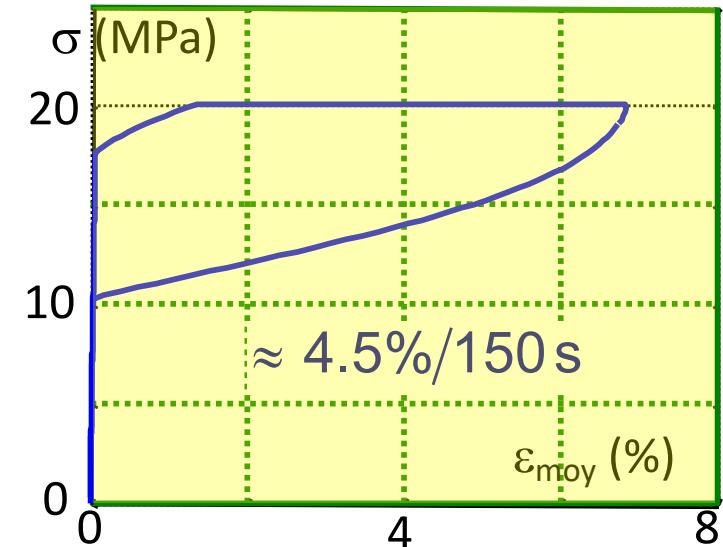
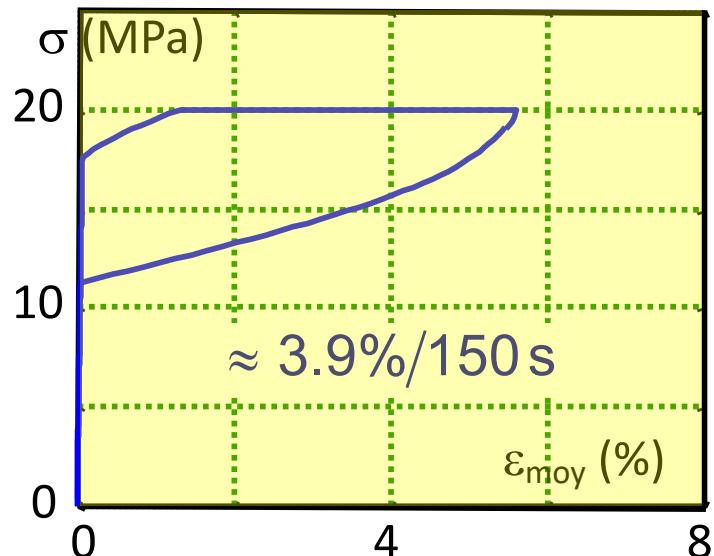
$$h_1 = 10 \text{ W/m}^2\text{K}$$

convection naturelle



$$h_2 = 60 \text{ W/m}^2\text{K}$$

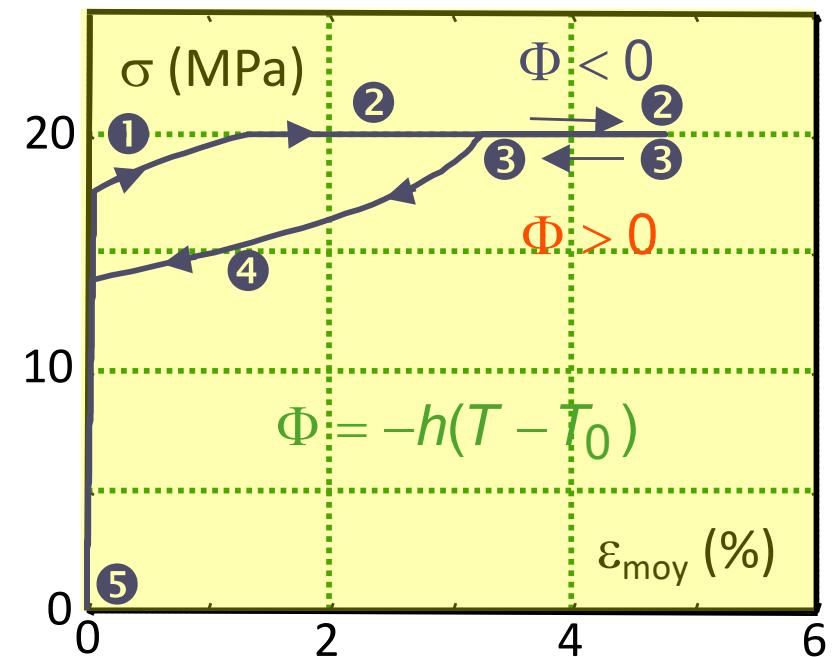
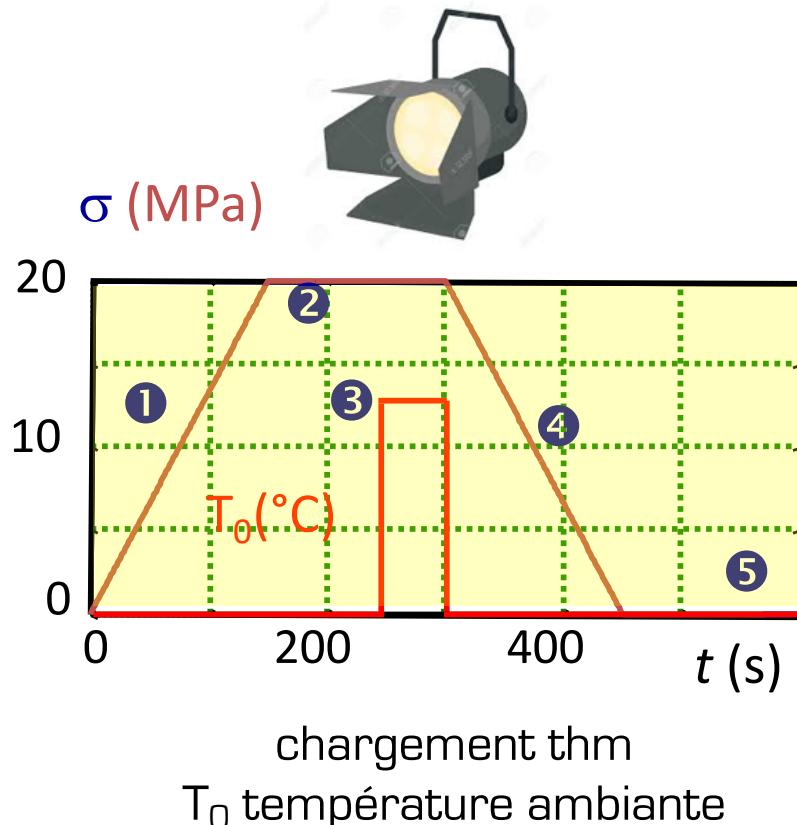
légère turbulence



fluage + hystérésis ... sans viscosité

Essai numérique (II)

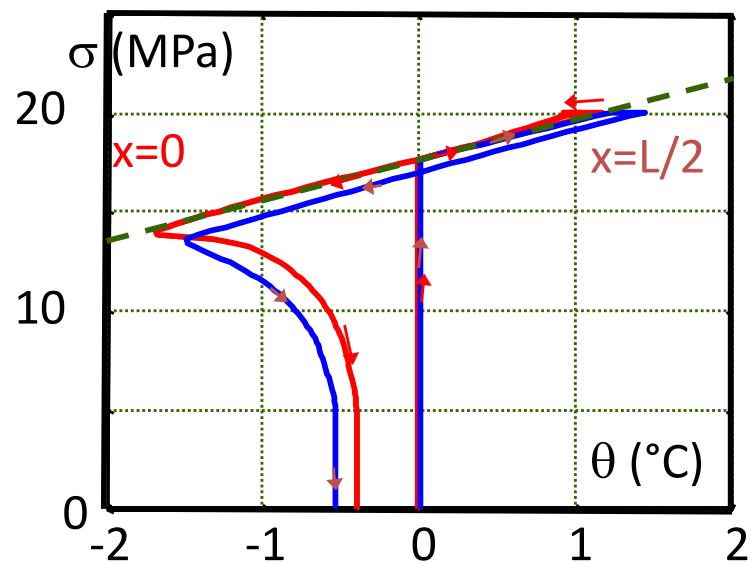
Fluage et fluage « inverse »



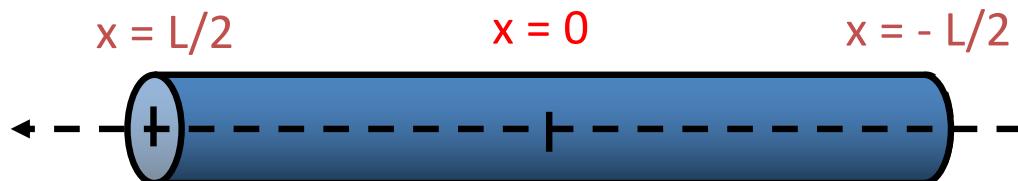
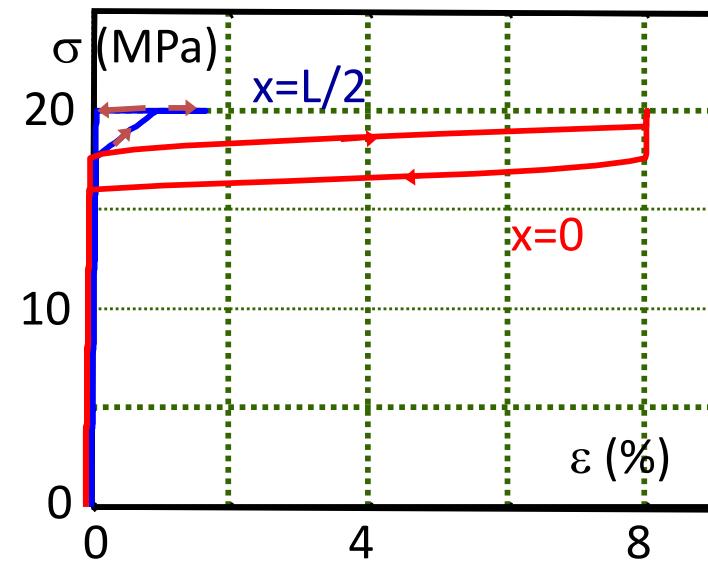
Sens du fluage induits par les échanges
de chaleur entre l'éprouvette et le milieu extérieur

Couplage + diffusion = effet d'échelle

Trajectoires thermomécaniques



Réponses mécaniques



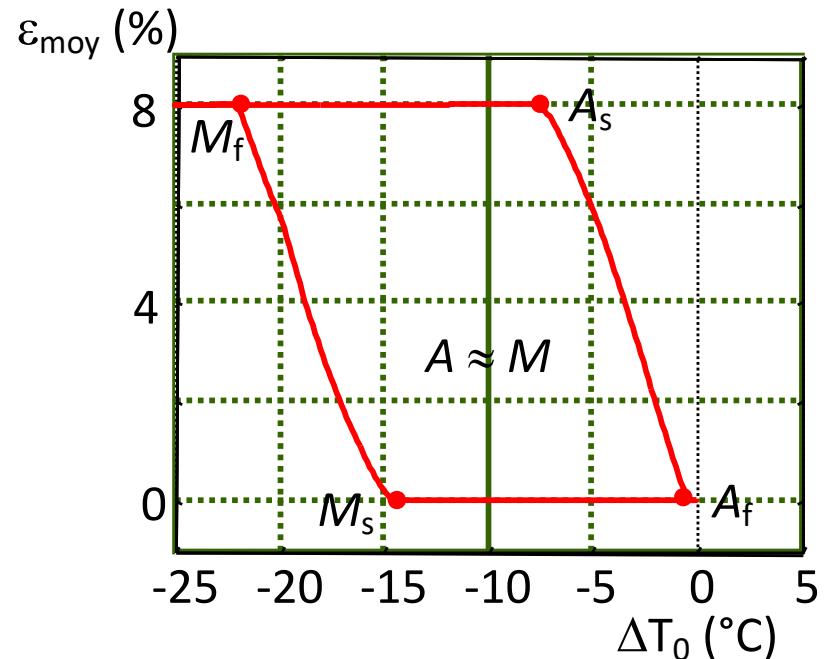
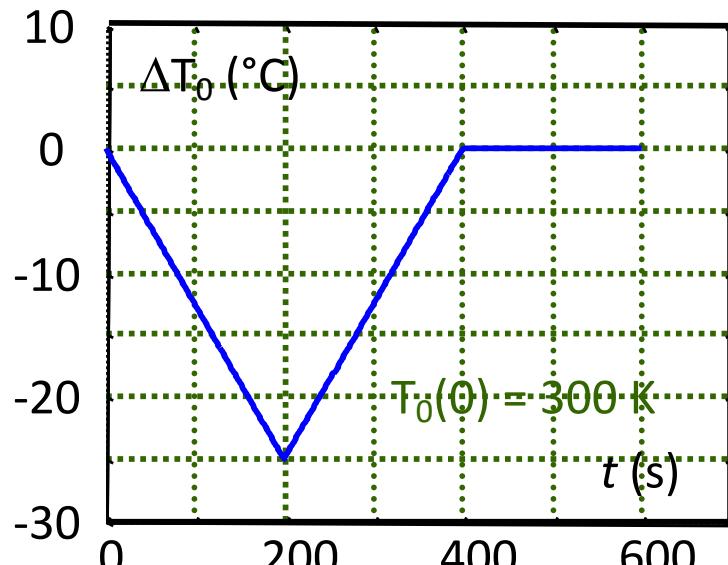
réponses locales \neq réponse globale
 identification des équations de comportement

Essai numérique (III)

Test de dilatométrie ($\sigma = \text{Cte}$)



Domaine de transition ($A - M = 10^{-3}^\circ\text{C}$)

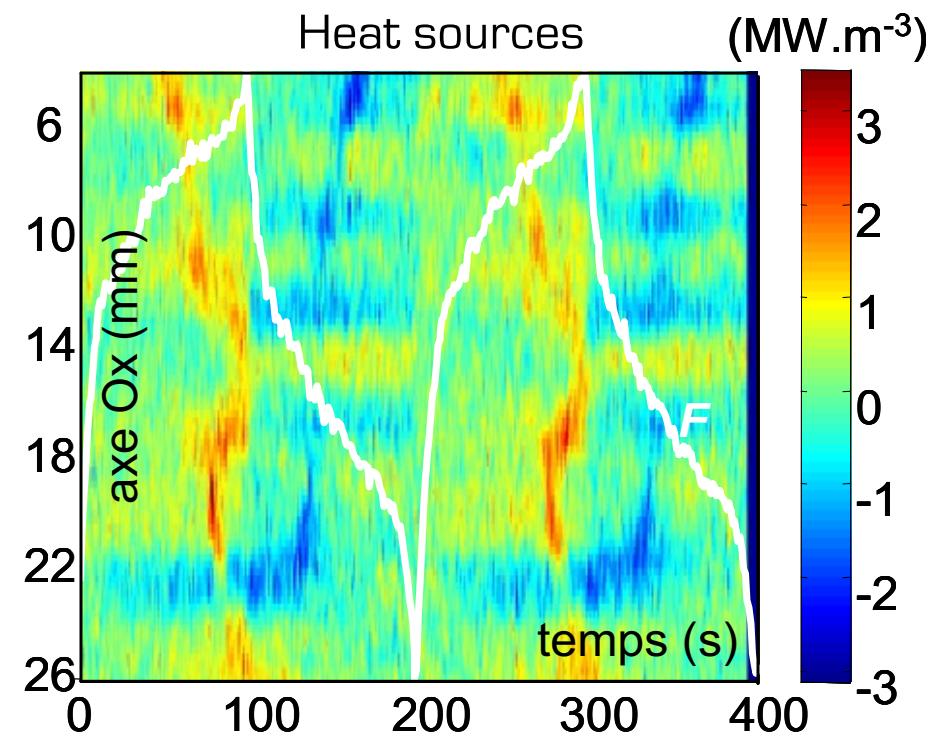
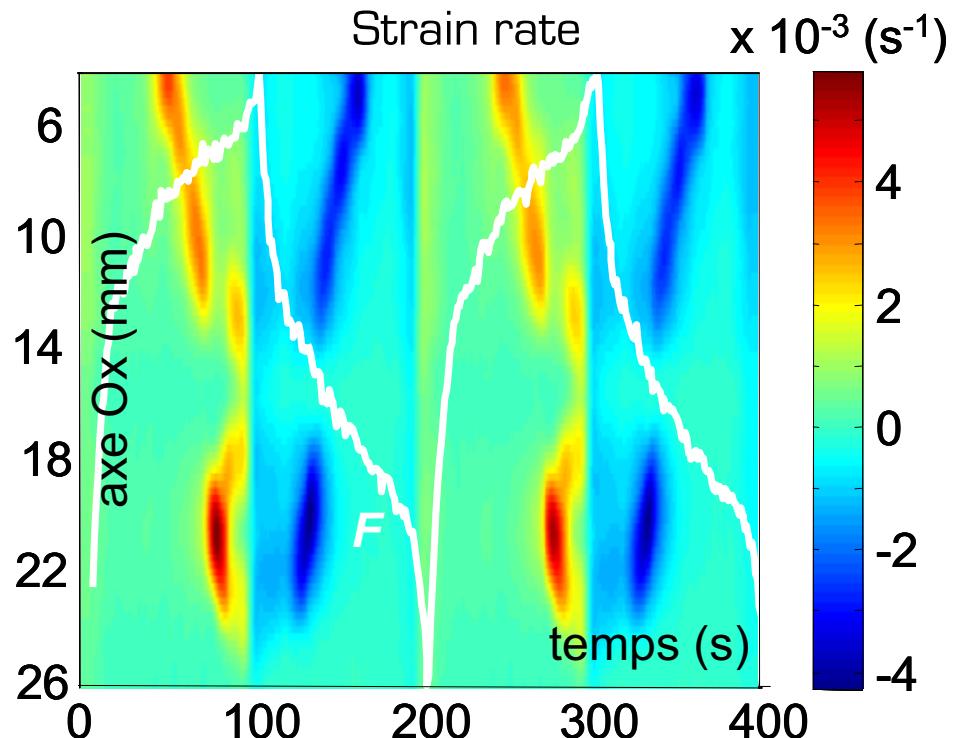


Diffusion de la chaleur, inertie thermique ...

$A - M = 10^{-3} \text{ }^\circ\text{C}$: domaine de transition du matériau

A_s, A_f, M_s, M_f : domaine de transition de l'échantillon

Load-unload cycles



Field of local responses (mesoscale)

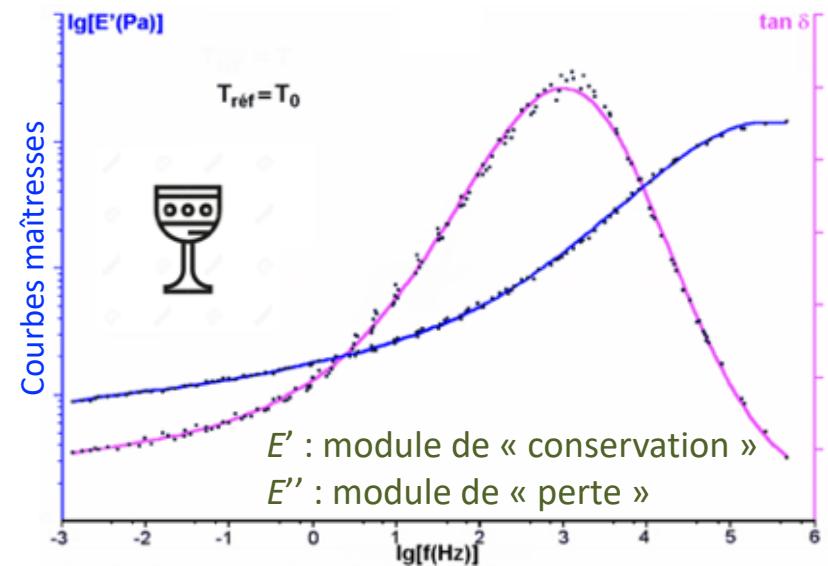
[S. Vigneron, PhD 09]



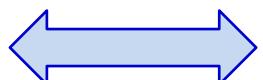
$$I_\varepsilon(\varphi) = \lim_{\eta \rightarrow 0^+} \int_{\mathbb{R}^n} V_\varepsilon(\eta) \, d\mu$$



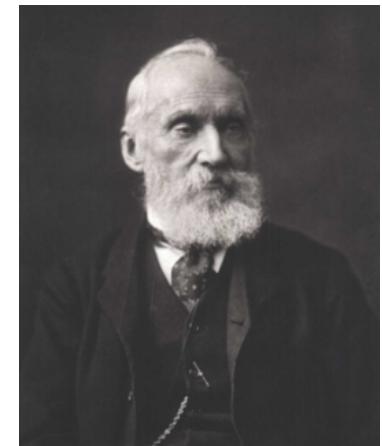
Effet du temps : viscosité ou
couplage fort et dissipation thermique



$$\begin{cases} \varepsilon = \frac{\sigma}{E} + \lambda_{th}(T - T_0) \\ \dot{T} + \frac{T - T_0}{\tau_{th}} = -\frac{E\lambda_{th}T\dot{\varepsilon}}{\rho C} \end{cases}$$



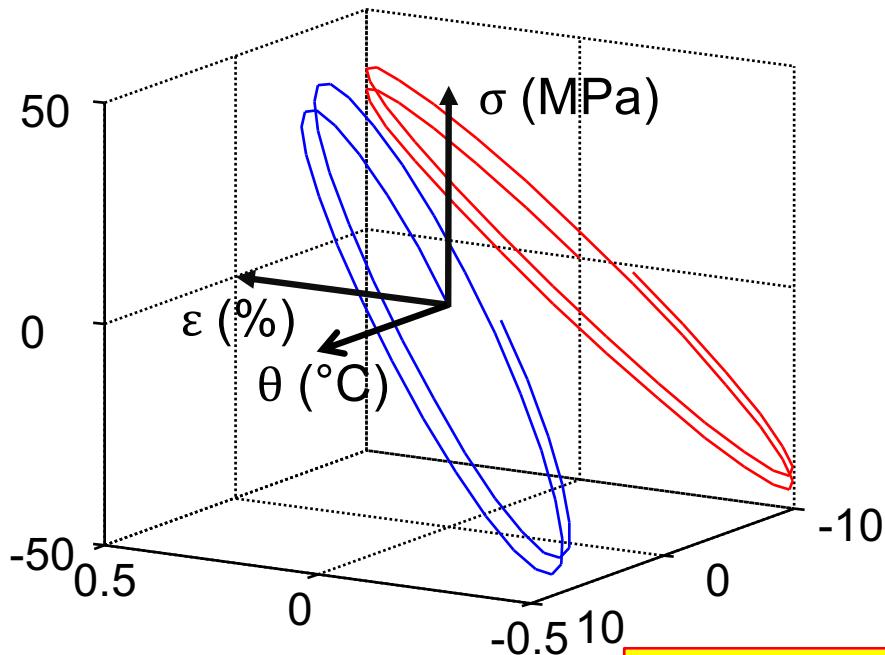
« OD » approach
linear heat losses



W. Thomson – Lord Kelvin
1824-1907

$E = 1000 \text{ MPa}$
 $\rho = 1000 \text{ kg.m}^{-3}$
 $C = 1000 \text{ J.kg}^{-1}.K^{-1}$

$\lambda_{th} = 50 \cdot 10^{-5} K^{-1} \times 100$
 $\tau_{th} = 30 \text{ s}$
 $T_0 = 294 \text{ K}$



Toward a
stabilised
cycle...

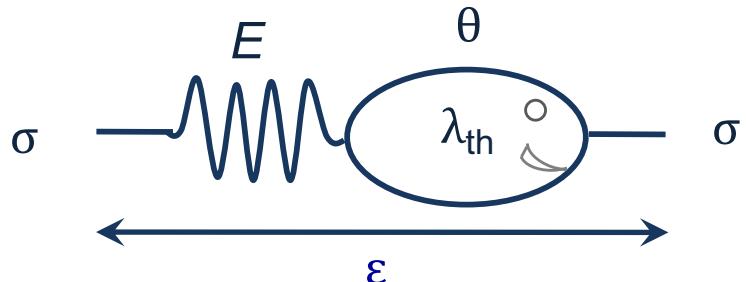
Thm couplings + thermal dissipation

$\tilde{W}_{def} = A_h = \tilde{W}_{the}$

Couplage ou dissipation ?

Couplage thermomélastique (i.e. $d_1=0$)

couplage fort



variables d'état (θ, ε)

$$\begin{cases} \varepsilon = \frac{\sigma}{E} + \lambda_{th} \theta \\ \dot{\theta} + \frac{\theta}{\tau_{th}} = - \frac{E \lambda_{th} (T_0 + \theta) \dot{\varepsilon}}{\rho_0 C_0} \end{cases}$$

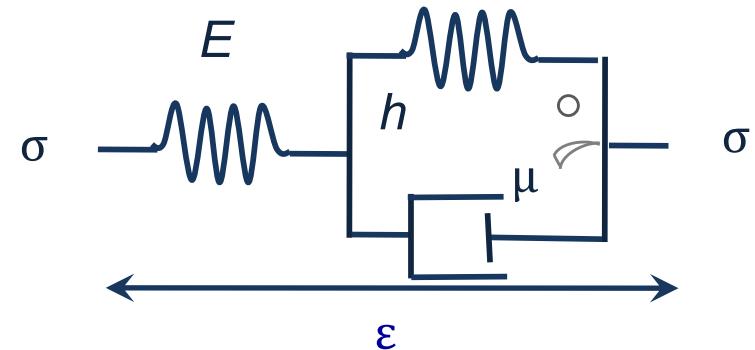
équation rhéologique

$$\sigma + \tau_{th} \dot{\sigma} \approx E \varepsilon + E \tau_{th} (1 + \chi) \dot{\varepsilon}$$



Processus dissipatif visqueux

couplage faible



variables d'état ($\varepsilon, \varepsilon_v$)

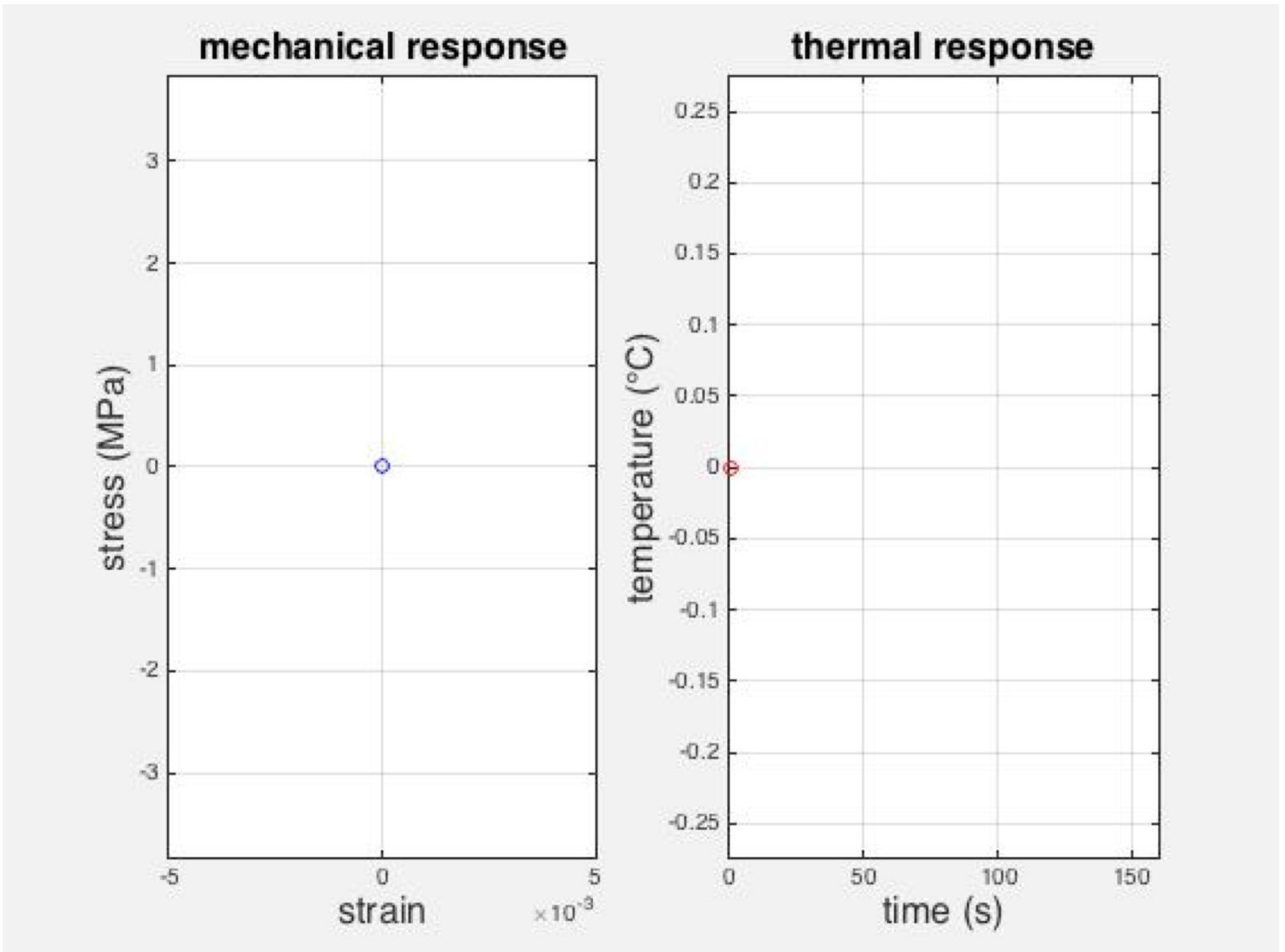
$$\begin{cases} \sigma = E(\varepsilon - \varepsilon_v) \\ \sigma = h \varepsilon_v + \mu \dot{\varepsilon}_v \end{cases}$$

équation rhéologique

$$\sigma + \frac{\mu}{E+h} \dot{\sigma} = \frac{Eh}{E+h} \varepsilon + \frac{E\mu}{E+h} \dot{\varepsilon}$$

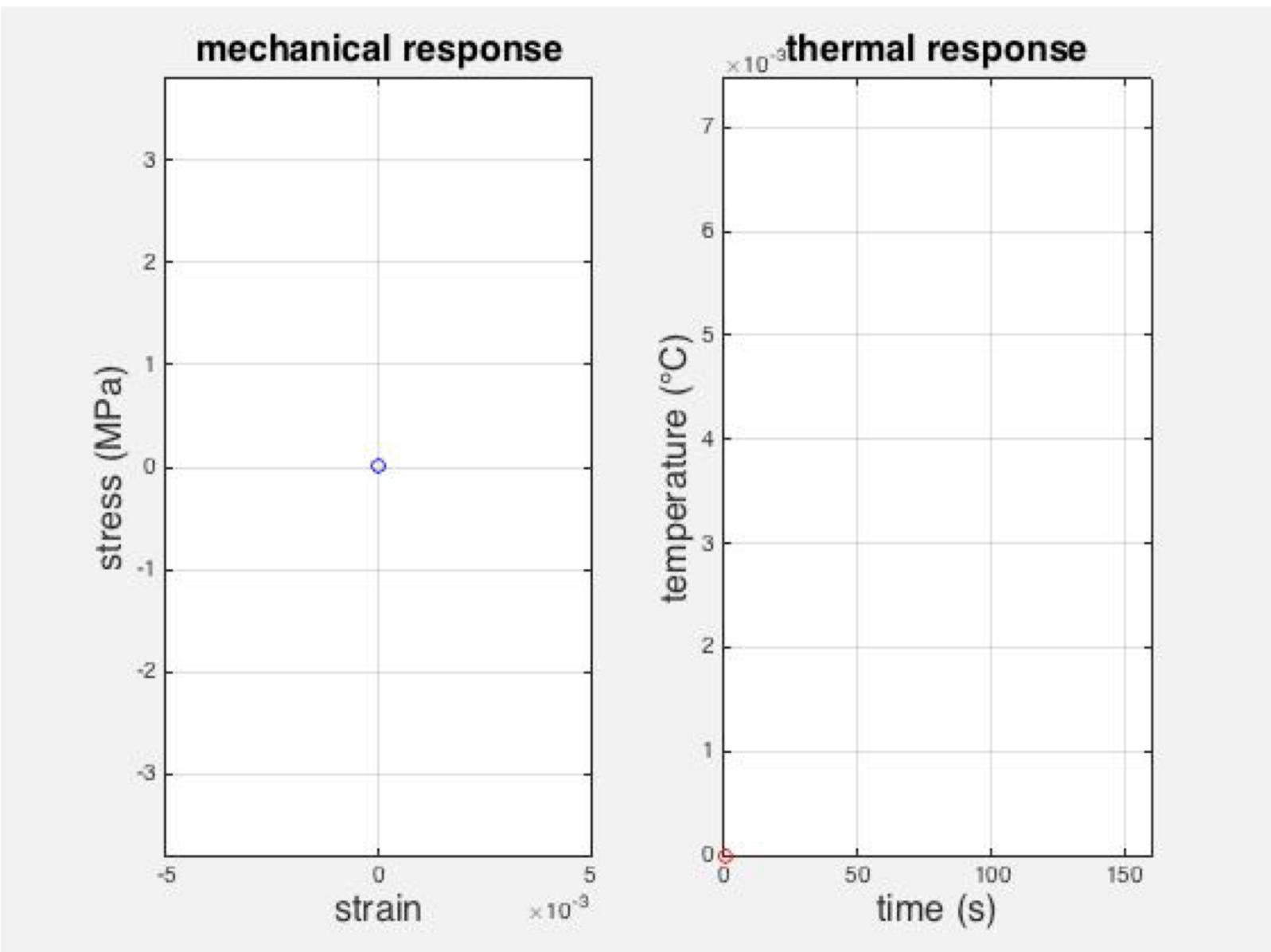
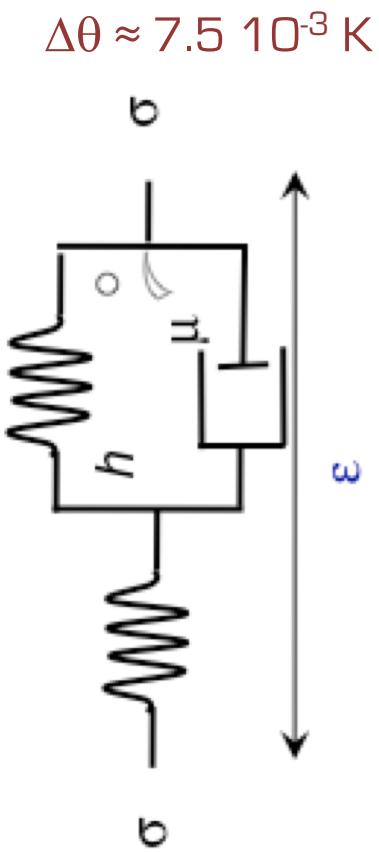
Visco-analyse de polymère (DMA)

Effets thermoélastiques : couplage fort

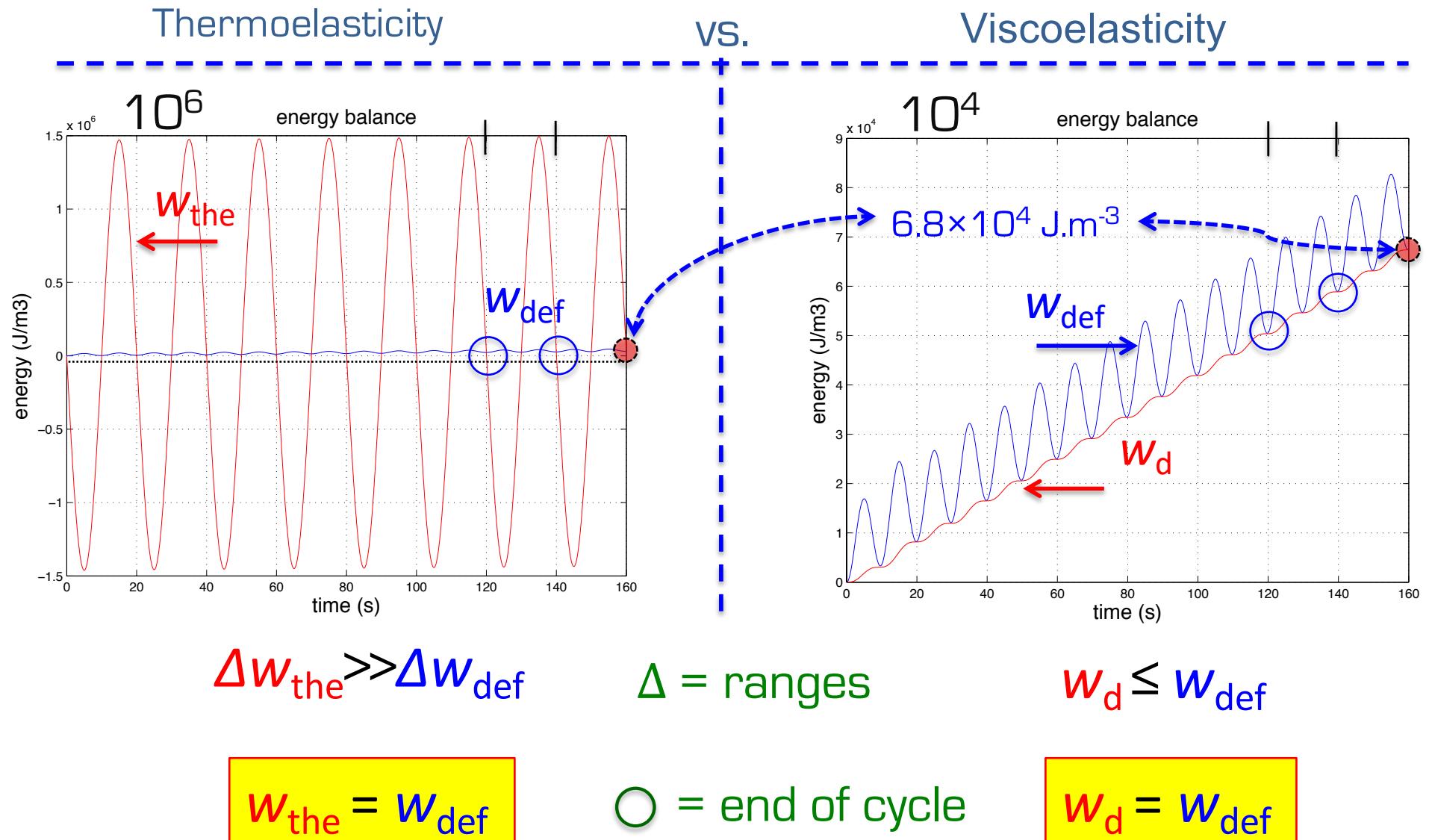


$$I_\varepsilon(\varphi) = \int_{\Omega} V(\varphi(x)) dx + \lim_{\substack{\leftarrow \\ \delta \rightarrow 0}} \int_{\Omega} \frac{1}{\delta} \int_{B_\delta(x)} F(u) dx$$

Effets visqueux : couplage faible



Energy balance : comparison

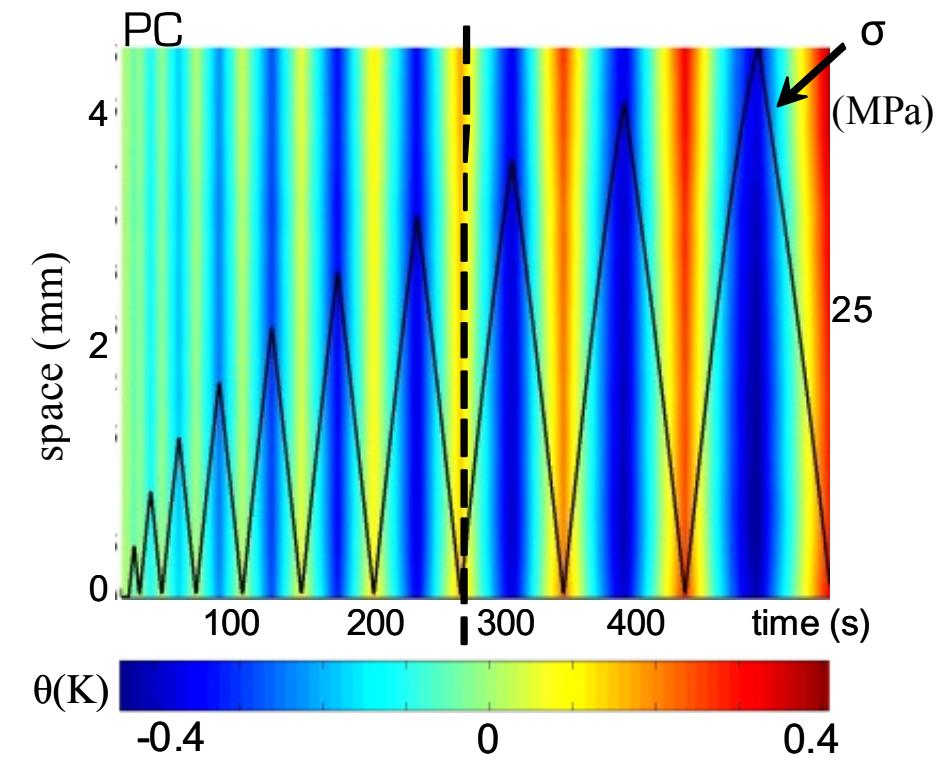
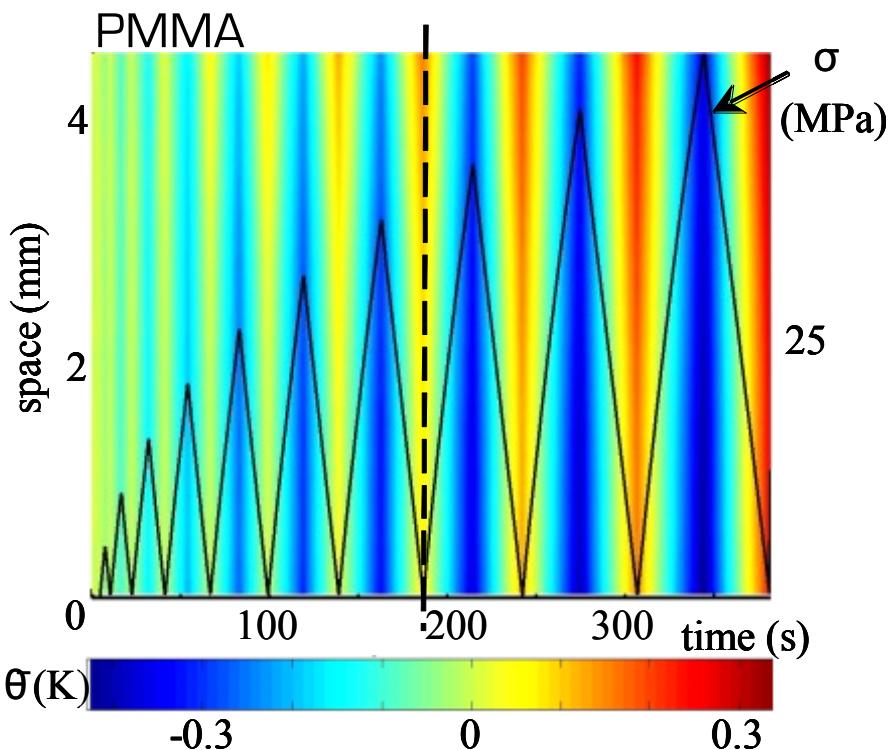


PMMA & PC : polymères à l'état vitreux

HPP isotherme : le royaume de la viscoélasticité linéaire

J. Alfrey (48), M. Biot (65), F. Sidoroff (70-75)

Equivalence of series and parallel models (e.g. P.T. and Z. models)

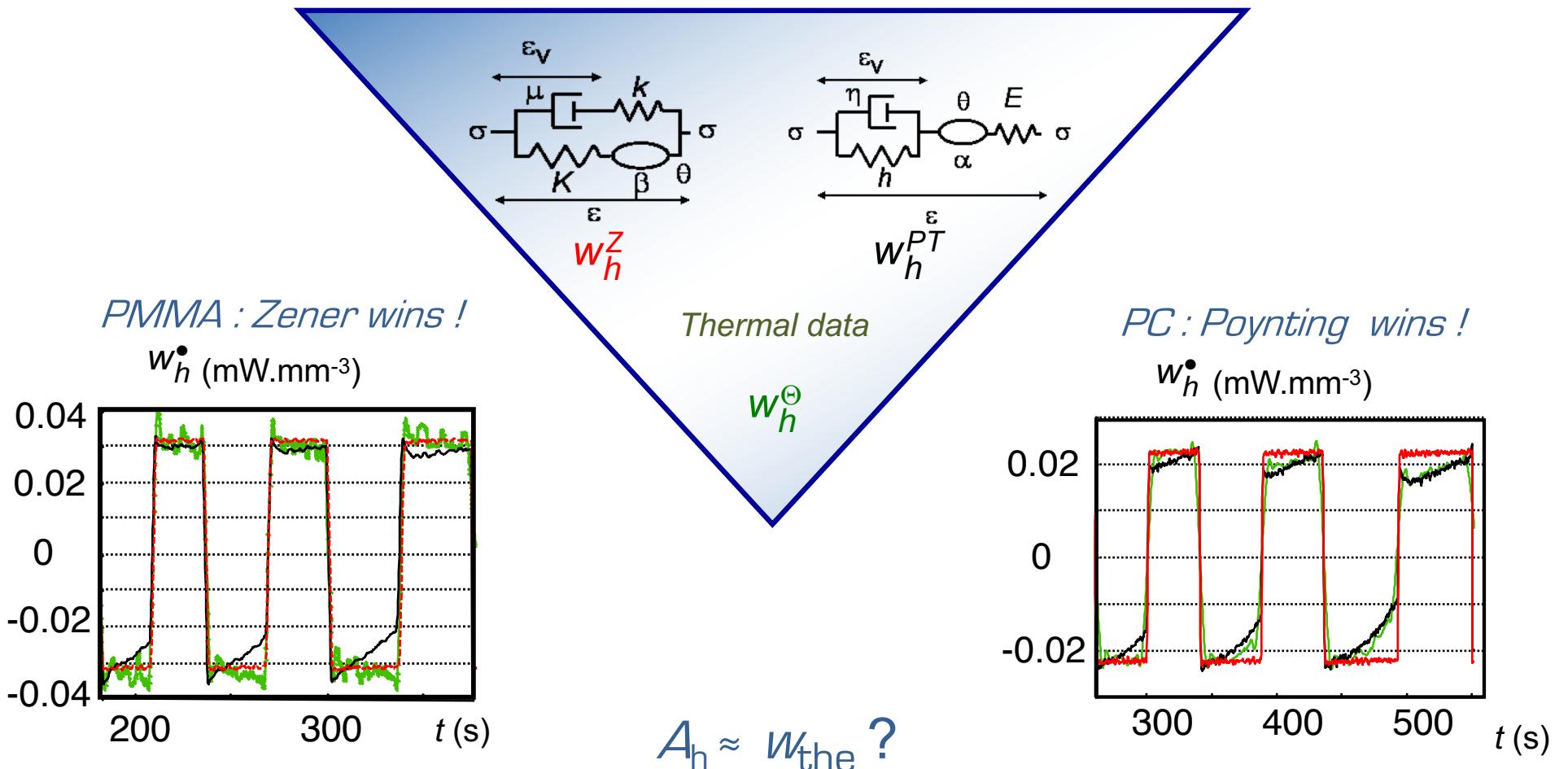


[S. Moreau, PhD 03]



PMMA and PC : simple models for glassy polymers

[PhD, Moreau S., CRAS 2005]

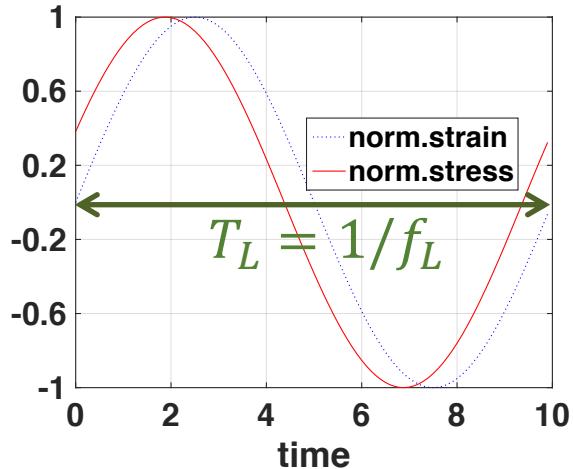


Low dissipative effects ! energy storage ?
Equivalence of series and parallel models ?

Stabilisation cyclique : polymères

Viscoélasticité linéaire : équivalence temps température (TTS - DMA)

Facteur de translation $a_{T_0}^T$



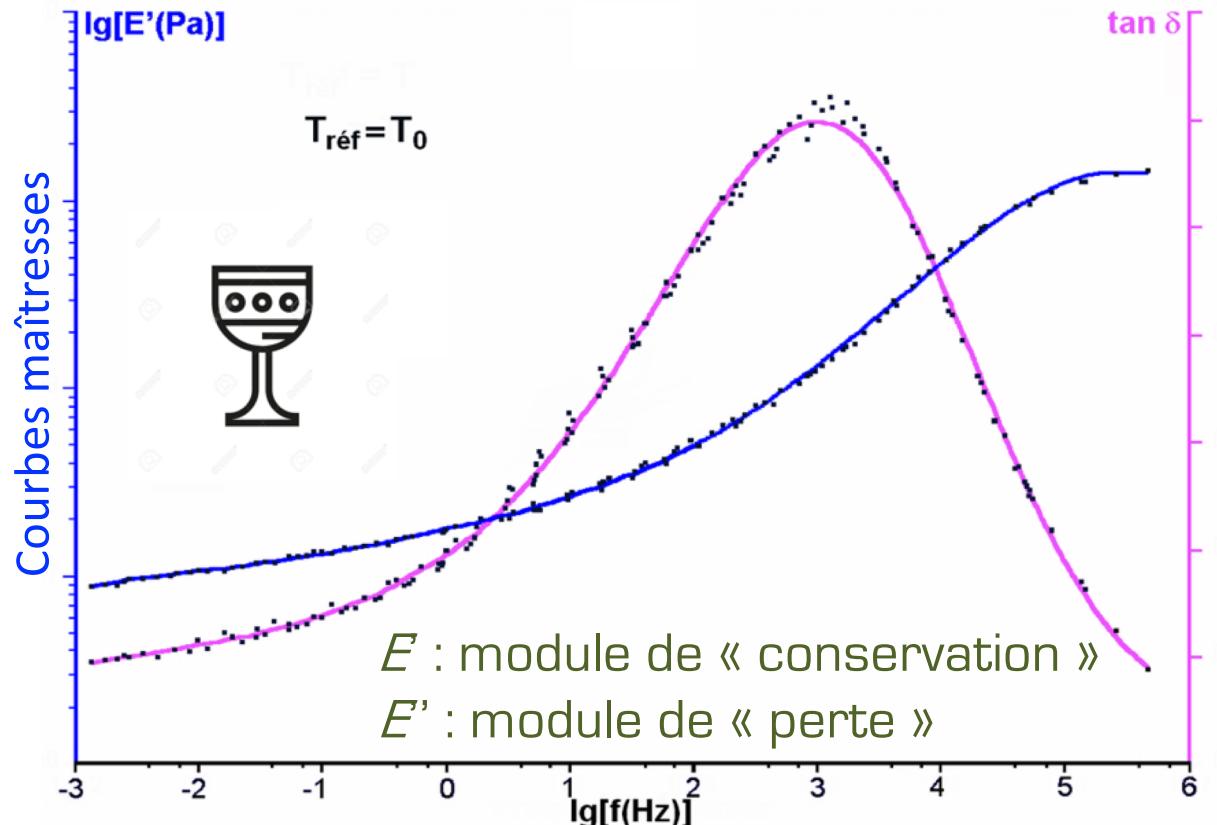
$$\varepsilon = \varepsilon_0 \sin(2\pi f_L t)$$

$$\sigma = \sigma_0 \sin(2\pi f_L t + \varphi)$$

$$\sigma = E' \varepsilon_0 \sin(2\pi f_L t) + E'' \varepsilon_0 \cos(2\pi f_L t)$$

$$\tan(\delta) = \frac{E''}{E'}$$

Pertes
Enr. Diss.



$$E'(f_L, T) = E'\left(f_L/a_{T_0}^T, T_0\right)$$

$$E''(f_L, T) = E''\left(f_L/a_{T_0}^T, T_0\right)$$

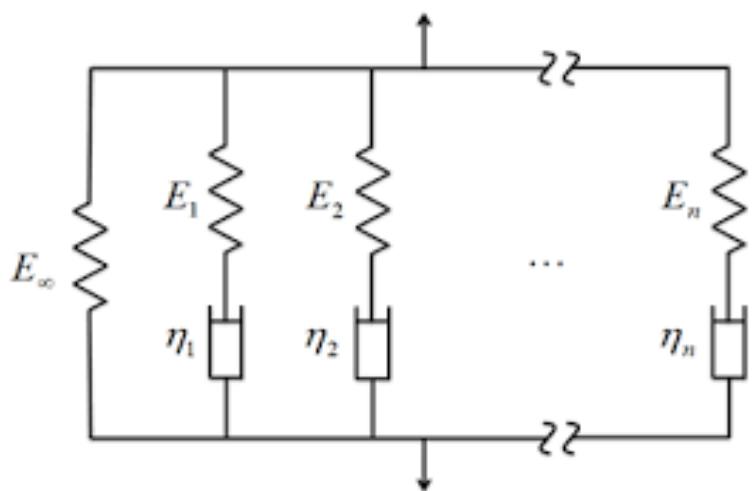
Analyse dynamique mécanique (TTS)

Idée : basse température + basse vitesse

haute température + haute vitesse

} « = » même mobilité moléculaire

Modèle de Maxwell généralisé
« anisotherme»



température = paramètre du modèle ...

$$\tau_i = \mu_i/E_i = \tau_i^0 f(T)$$

Quelques ingrédients théoriques :

$$E' = E_\infty + \sum_1^N \frac{E_i \tau_i^2 \omega^2}{1 + \tau_i^2 \omega^2}$$

$$E'' = \sum_1^N \frac{E_i \tau_i \omega}{1 + \tau_i^2 \omega^2}$$

Arrhenius [ou WLF ou ...]

$$\tau_i = \tau_i^0 \exp\left(-\frac{E_a}{R}\left(\frac{1}{T} - \frac{1}{T_0}\right)\right)$$

La température est
une variable contrôlée ($T = T_{DMA}$)

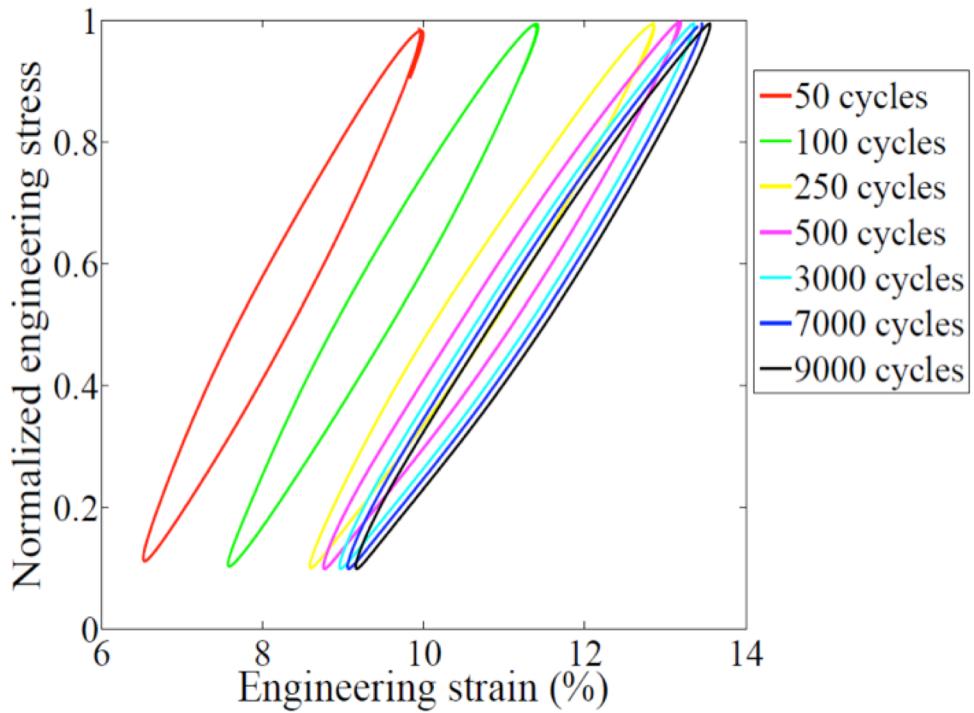
Pas de couplage, pas d'auto-échauffement !

Wet polyamide 6.6 (I)

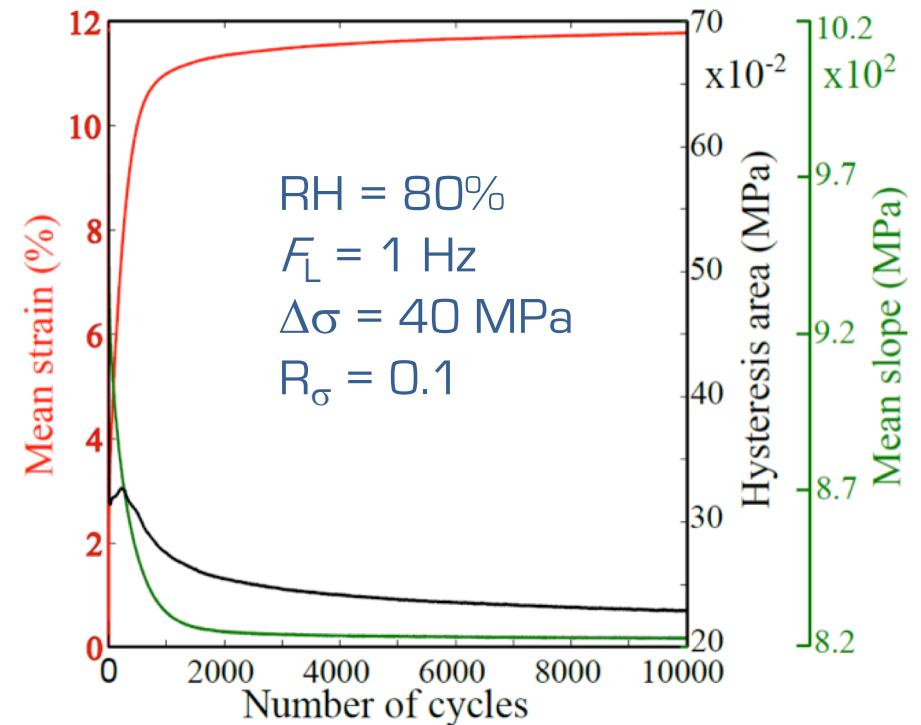
[A. Benaarbia et al., MoM, 16]

Overall mechanical responses

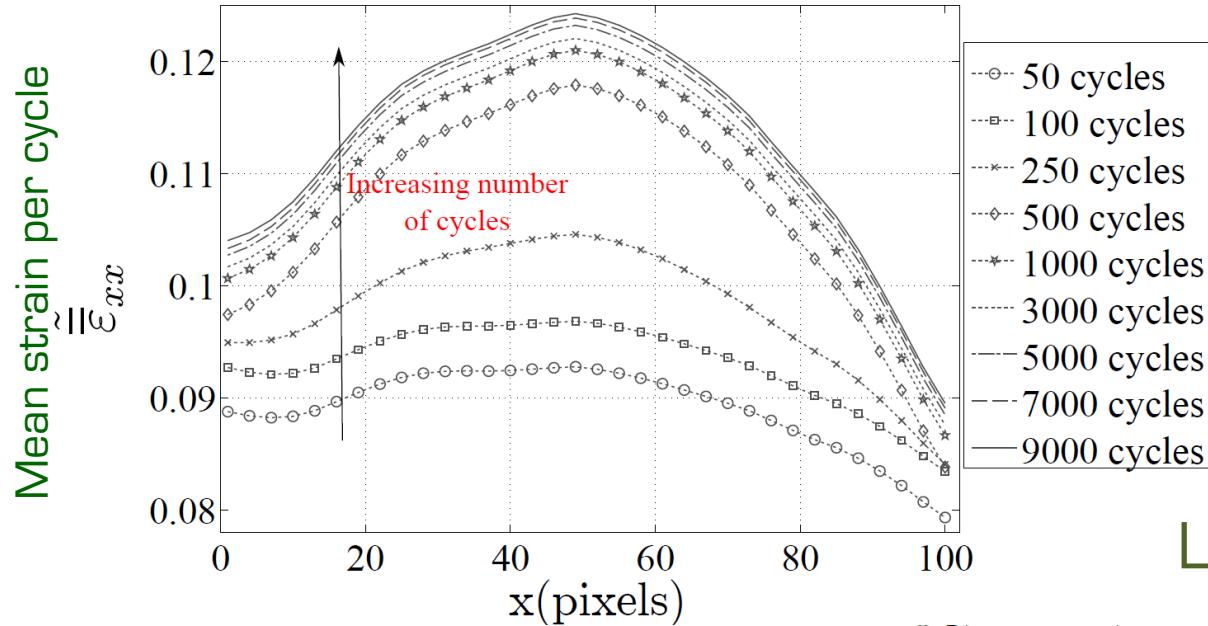
Stress-strain loops



Loop parameters

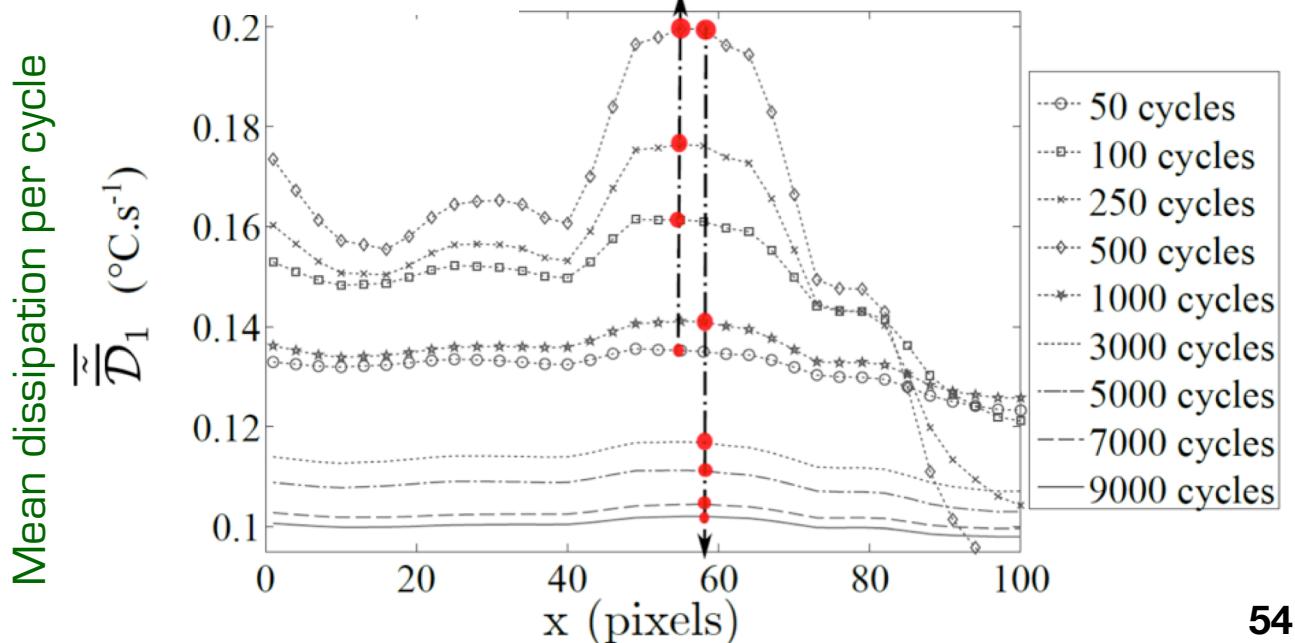


Wet polyamide 6.6 (II)



Strain concentration
 $[0,500]$
 Progressive stabilization
 $[1000,9000]$

Longitudinal profiles

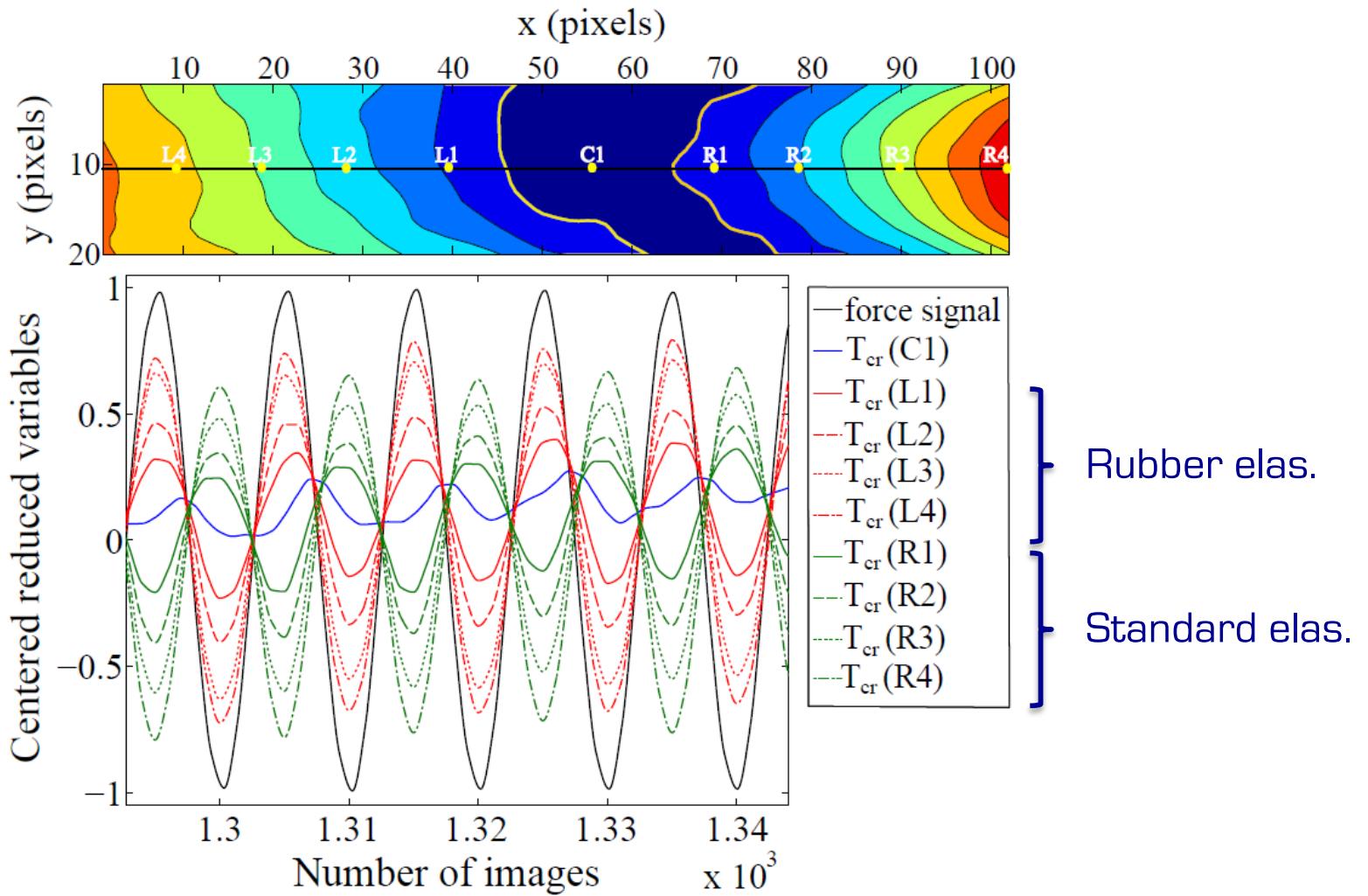


Localization of dissipation
 $[0,500]$
 Progressive stabilization
 $[1000,9000]$

Wet polyamide 6.6 (III)

[A. Benaarbia et al., MoM, 16]

Local “thermoelastic” temperature ranges



Wet polyamide 6.6 (IV)

Comments



- ✓ **Self heating** induced by dissipation (20 °C)
- ✓ Glass transition temperature Tg (close to RT !)
- ✓ From glassy material to rubbery material ??
- ✓ **Water plasticizing** ($T_g \downarrow$ when RH \uparrow)
- ✓ Chemical ageing
- ✓ Thermo-hydro-chemo-mechanical modeling
- ✓ **Role of glass fibers** ... multiscale approach
- ✓ Structures designed for **fatigue**



Cemef



LEM3
LABORATOIRE D'ÉTUDE DES MICROSTRUCTURES
ET DE MÉCANIQUE
DES MATERIAUX